



HAL
open science

Correct determination of charge transfer state energy from luminescence spectra in organic solar cells

Mathias List, Tanmoy Sarkar, Pavlo Perkhun, Jörg Ackermann, Chieh Luo,
Uli Würfel

► **To cite this version:**

Mathias List, Tanmoy Sarkar, Pavlo Perkhun, Jörg Ackermann, Chieh Luo, et al.. Correct determination of charge transfer state energy from luminescence spectra in organic solar cells. *Nature Communications*, 2018, 9 (1), 10.1038/s41467-018-05987-8 . hal-01901500

HAL Id: hal-01901500

<https://amu.hal.science/hal-01901500>

Submitted on 22 Oct 2018

HAL is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.



Distributed under a Creative Commons Attribution 4.0 International License

ARTICLE

DOI: 10.1038/s41467-018-05987-8

OPEN

Correct determination of charge transfer state energy from luminescence spectra in organic solar cells

Mathias List¹, Tanmoy Sarkar¹, Pavlo Perkhun², Jörg Ackermann², Chieh Luo¹ & Uli Würfel ^{1,3}

Generation and recombination of electrons and holes in organic solar cells occurs via charge transfer states located at the donor/acceptor interface. The energy of these charge transfer states is a crucial factor for the attainable open-circuit voltage and its correct determination is thus of utmost importance for a detailed understanding of such devices. This work reports on drastic changes of electroluminescence spectra of bulk heterojunction organic solar cells upon variation of the absorber layer thickness. It is shown that optical thin-film effects have a large impact on optical out-coupling of luminescence radiation for devices made from different photoactive materials, in configurations with and without indium tin oxide. A scattering matrix approach is presented which accurately reproduces the observed effects and thus delivers the radiative recombination spectra corrected for the wavelength-dependent out-coupling. This approach is proven to enable the correct determination of charge transfer state energies.

¹Fraunhofer Institute for Solar Energy Systems (ISE), Heidenhofstr. 2, 79110 Freiburg, Germany. ²Centre Interdisciplinaire de Nanoscience de Marseille (CINaM), Campus de Luminy, Case 913, 13288 Marseille, France. ³Materials Research Center, University of Freiburg, Stefan-Meier-Str. 21, 79104 Freiburg, Germany. Correspondence and requests for materials should be addressed to U.W. (email: uli.wuerfel@ise.fraunhofer.de)

Organic photovoltaics (OPV) has the potential of low-cost and large-area production of flexible and light-weight solar cells. However, the power conversion efficiencies (PCE) achieved thus far are not yet competitive to commercially available technologies like, e.g., crystalline silicon solar cells. Hence, improving the PCE is a key challenge in OPV research and development. In the commonly used organic bulk hetero-junction solar cell (OSC), a charge transfer (CT) complex is formed at the donor–acceptor (D–A) interface due to interaction of the D and A molecules, resulting in an absorption and emission band at lower energies compared to the optical gaps of the individual D and A materials. Although the CT complex does not contribute significantly to the overall absorption in OSC it nevertheless is the intermediate state over which charge carrier generation and recombination occur forming a CT exciton^{1–4}. The term CT state will be used here to refer to the excited state of the CT complex where the electron is located at the acceptor and the hole at the donor where the CT energy (E_{CT}) is the excitation energy with respect to the CT complex ground state. The energy (E_{CT}) and occupation of the CT states are crucial as they determine the open-circuit voltage (V_{OC}) and are thus directly correlated to the PCE^{1,5–7}, which will be discussed in more detail below.

Luminescence spectroscopy is used to investigate CT states^{6,8–13}; however, the quantum yield of radiative CT recombination of typically used D–A blends in OSC lies in the range of 10^{-9} – $10^{-66,14}$ and is thus not trivial to detect. Especially, electroluminescence (EL) spectroscopy is suitable to examine CT complexes since EL is dominated by CT state emission^{6,15}. Therefore, this work will focus on EL, while the results are valid for any kind of radiative emission, such as photoluminescence.

For organic light-emitting diodes (OLED), microcavity effects have been reported to influence emission^{16–18} and optical out-coupling is a central part of the conducted research^{19,20}. Optical simulations in the field of OPV are commonly used to improve wavelength-dependent optical field distribution (incoupling of sunlight) and therefore generation^{21–23}. Microcavity effects have been used to enhance quantum efficiency close to the absorption band edge and thus improve device performance^{24–29} and they were furthermore utilized to alter transmission and produce color-tunable semitransparent OSCs in the visible range³⁰. As a matter of fact, these OPV-related optical studies focus on the wavelength regions of the donor and acceptor absorption and thus ultimately on the optimization of the external quantum efficiency (EQE). In contrast, to the knowledge of the authors microcavity effects have not been considered yet when using luminescence spectroscopy to characterize radiative recombination occurring via CT states within OSCs, i.e., at significant longer wavelengths. For this reason, this work focuses on the wavelength region of CT state absorption and emission in OSC and takes optical out-coupling effects explicitly into account. It will be shown that it is essential to account for the optical properties of the investigated system when CT state emission is detected and analyzed by luminescence spectroscopy.

Absorption and emission spectra of CT states are Stokes shifted due to the reorganization of the CT complex³¹. According to Marcus theory, the reduced CT state emission spectrum I_f/E (where I_f is the emission spectra and E the energy) measured in photons per unit time and unit energy^{1,32} is given by:

$$\frac{I_f}{E} = \frac{f_{I_f}}{\sqrt{4\pi\lambda_{RO}k_B T}} \exp\left(\frac{-(E_{CT} - \lambda_{RO} - E)^2}{4\lambda_{RO}k_B T}\right), \quad (1)$$

with λ_{RO} being the reorganization energy, f_{I_f} being a prefactor depending on the electronic coupling matrix element of the D–A system, k_B the Boltzmann constant, and T the temperature. E_{CT}

can be determined by fitting Eq. (1) to the experimentally determined EL spectra. The peak position for CT state emission is therefore expected to be $E_{CT}^{em} = \max(I_f/E) = E_{CT} - \lambda_{RO}$, i.e., red-shifted by λ_{RO} . In the same manner, CT state absorption spectra will be blue shifted leading to a characteristic Stokes shift $2\lambda_{RO}$ of ~ 0.4 eV².

Under open-circuit conditions, generation and recombination are equally large which determines the electron and hole concentrations. The open-circuit voltage of a solar cell is given by:

$$eV_{OC} = E_G - k_B T \ln\left(\frac{N_{ETL}N_{HTL}}{n_e n_h}\right) \quad (2)$$

Therein, N_{ETL} and N_{HTL} are the effective densities of electronic states of the electron and hole transport levels (ETL in the acceptor and HTL in the donor) and e the elementary charge. For conventional semiconductors, E_G resembles the band gap. In OSC, an effective gap $E_{G,eff}$ is considered as the energy difference between electrons in the ETL and holes in the HTL.

Accounting for energetic disorder and under the assumption of thermodynamic equilibrium between free charge carriers and the occupation of CT states, Burke et al. derived the relation

$$eV_{OC} = \underbrace{E_{CT} - \frac{\sigma_{CT}^2}{2k_B T}}_{E_{G,eff}} - k_B T \ln\left(\frac{efN_0L}{\tau_{CT}J_{rec}}\right) \quad (3)$$

and verified it to be valid for OSC⁷. For this, a Gaussian-shaped energetic CT state distribution with standard deviation σ_{CT} and center E_{CT} is assumed representing the inhomogeneously broadened CT state energies at the D–A interfaces. Here, $\tau_{CT} = 1/k_r$ is the average lifetime of the CT states, J_{rec} the recombination current ($J_{rec} \approx J_{SC}$) under open-circuit conditions, f is the volume fraction of the D–A interfaces, N_0 is the density of CT states related to the interfacial volume, and L the thickness of the photoactive layer (PAL). In case of a Gaussian energetic disorder, the experimentally observed CT spectrum is shifted to $E_{CT}^{exp} = E_{CT} - \sigma_{CT}^2/2k_B T$ with $\lambda_{RO}^{exp} = \lambda_{RO} + \sigma_{CT}^2/2k_B T$. From Eq. (3), it can be seen that this experimentally observed CT state energy $E_{CT}^{exp} = E_{CT} - \sigma_{CT}^2/2k_B T$ resembles exactly the effective band gap $E_{G,eff}$ for OSC. Gaussian energetic disorder leads to a reduction of V_{OC} by $\sigma_{CT}^2/2k_B T$ and thus depends on the width of the disorder distribution. Note that the energy of a particular CT complex depends on the local environment at the D–A interface, which means that not all charge carriers can thermalize down to the low energetic tail of the distribution. From Eqs. (1) and (3), it is obvious that knowledge about CT state energy E_{CT} and the width of emission or absorption spectra are key for a detailed understanding and thus optimization of V_{OC} in OSC. Here we show that interference effects have a significant impact on emission spectra in the relevant range and have to be taken into account when CT state energies are determined by luminescence spectroscopy.

Results

Optical simulation of radiative out-coupling. Emitted electromagnetic waves are partially reflected at material interfaces in multilayer structures due to differences in their optical coefficients, i.e., refractive indices (n) and extinction coefficients (k). The direct and reflected waves superimpose leading to constructive and destructive interference. This can alter the wavelength dependency of the spectrum measured outside the device $I(\lambda)$ in comparison to the intrinsic emission line shape $I_{hom}(\lambda)$ of the emitting material, i.e., the emitted spectrum in absence of any reflecting interfaces. The radiative out-coupling factor γ_{OC} is

defined as the quotient of these two quantities:

$$\gamma_{\text{OC}}(\lambda) = \frac{I(\lambda)}{I_{\text{hom}}(\lambda)} \quad (4)$$

As the emitted radiation in the organic solar cells investigated in this study originates from the entire PAL $\bar{\gamma}_{\text{OC}}$ is defined as the average out-coupling factor over all emission positions within the PAL as outlined in detail below. For simulations of the radiance enhancement within the active layer of the OSCs, an implementation³³ of the scattering matrix (S matrix) method^{34,35} based on custom code was applied. Within this method, the exact configuration of the investigated OSC layer stacks can be simulated. Experimentally determined n and k values are used as input parameters. All used parameters can be found in the Supplementary Fig. 1, 2. Super- and substrate are air ($n = 1$) and glass ($n = 1.67$), respectively, or vice versa depending on the layer stack. When the luminescence is observed through the 1.1 mm thick glass substrates, the latter was modeled as superstrate with infinite elongation. Thereby, no reflection at the glass–air interface is included in the simulation, i.e., the back reflection at the 1.1 mm distant glass–air boundary is assumed to be not coherent and does not contribute to interference effects within the stack.

The used layer stacks are one-dimensional photonic structures, therefore a one-dimensional simulation along the direction perpendicular to the electrodes (referred to as x -axis) fully describes these systems. Luminescence within the photoactive layer (PAL) was implemented by a dipole-like emitter and its position x was varied throughout the whole PAL. For this, the layer stack was split at position x and two S matrices, one for the lower and one for the upper layers, were calculated. The out-coupled radiance $I_{\text{PAL}}(x)$ from the device into the superstrate for each dipole position was obtained from the time-averaged Poynting vector $\bar{\mathbf{S}}$ with a sampling of $\Delta x = d_{\text{PAL}}/N$, with d_{PAL} being the thickness of the photoactive layer. The error was proven to be <1% for $N \geq 100$ (relative to $N = 1000$), thus $N = 100$ was used for the results shown in this work. To obtain a spherically symmetrical emission the dipole was oriented in x -, y -, and z -direction successively and the resulting radiances were averaged. Under the assumption of a relative permeability of $\mu_r = 1$ for electromagnetic plane waves $|\bar{\mathbf{S}}|$ can be calculated by the amplitude $|\mathbf{E}|$ of the electric field:³⁶

$$|\bar{\mathbf{S}}| = \frac{1}{2} \sqrt{\frac{\epsilon_0 \epsilon_r}{\mu_0}} |\mathbf{E}|^2 \quad (5)$$

with ϵ_0 , ϵ_r , and μ_0 being the vacuum and relative permittivity and the permeability of free space, respectively. As reference, the free space radiance I_{hom} of a homogeneous system with infinite elongation was calculated. To obtain the radiative out-coupling factor $\gamma_{\text{OC}}(x)$ the radiance $I_{\text{PAL}}(x)$ for an emission from the full stack was divided by the reference radiance I_{hom} ³⁷ according to Eq. (4). Assuming spatially homogeneous recombination and thus homogeneous luminescence within the PAL of an OSC, the radiative out-coupling factor of the system was calculated as spatial average of $\gamma_{\text{OC}}(x)$, i.e., the spatial average over all emission positions x in the PAL:

$$\bar{\gamma}_{\text{OC}} = \frac{1}{d_{\text{PAL}}} \int_0^{d_{\text{PAL}}} \gamma_{\text{OC}}(x) dx = \frac{1}{d_{\text{PAL}}} \int_0^{d_{\text{PAL}}} \frac{I_{\text{PAL}}(x)}{I_{\text{hom}}} dx \quad (6)$$

Since γ_{OC} and thus $\bar{\gamma}_{\text{OC}}$ are relative quantities, the amplitude of the dipole emitter can be chosen arbitrarily and was set to unity.

Luminescence spectroscopy. To investigate CT state emission, EL spectroscopy was performed for a variation of photoactive D–A systems and layer stacks. The EL spectrum $I_{\text{EL}}(\lambda)$ measured in power per unit area and per unit wavelength for a device with a PAL consisting of P3HT:PC₆₁BM with PEDOT:PSS as transparent electrode (device type A in Fig. 1a is plotted as blue solid line in Fig. 2). The emission features a rather wide peak with a maximum at ~ 1360 nm.

After measuring the EL spectrum, the very same device (type A) was modified by depositing a 10 nm thick (and thus semitransparent) Ag layer on top of the transparent electrode and renamed to type B, depicted in Fig. 1b. When measuring EL again, drastic changes are observed in the spectrum which can be seen from the solid red line in Fig. 2. It shows a narrower peak with a central wavelength of about 1250 nm. It should be pointed

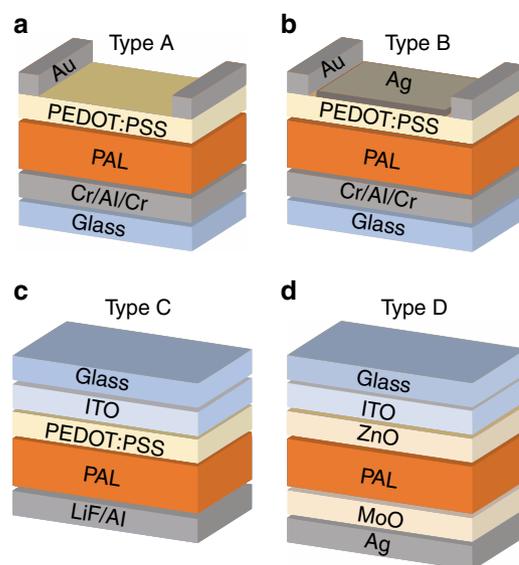


Fig. 1 Schematic illustration of device architectures of the used organic solar cells. Layer stacks of an optically inverted OSC (a, b) with additional semitransparent Ag layer. The ITO-based architecture using PEDOT:PSS and LiF/Al electrodes is shown in c and inverted ITO-based architecture using ZnO and MoO₃ as charge selective layers in d

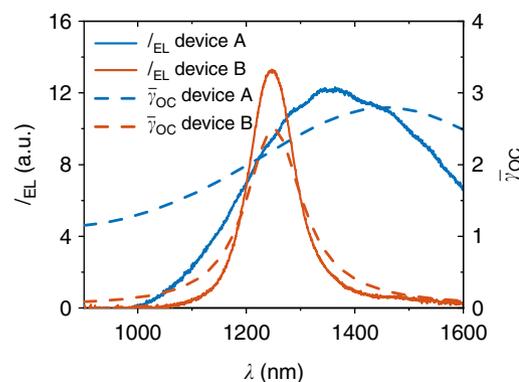


Fig. 2 EL spectra and out-coupling factor of an OSC with and without semitransparent layer. EL spectra of a P3HT:PC₆₁BM cell without (type A, see Fig. 1) and with a thin, semitransparent Ag layer on top (type B) together with the calculated radiant out-coupling factors $\bar{\gamma}_{\text{OC}}$. The more distinct peak with the additional semitransparent layer originates from the formation of an optical cavity what can be seen from the out-coupling factor

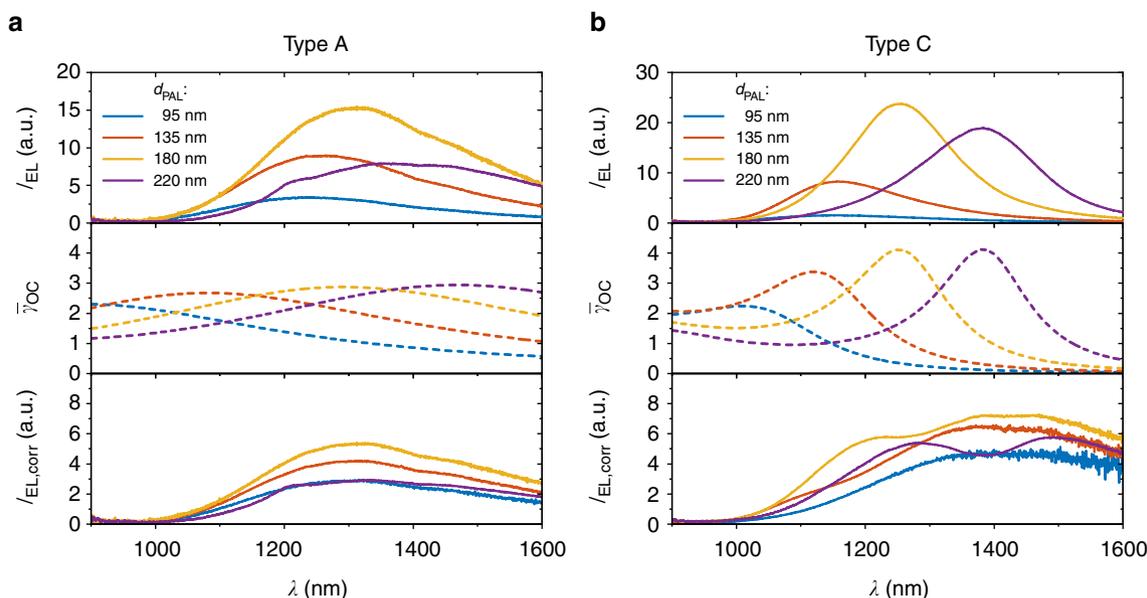


Fig. 3 EL spectra $I_{EL}(\lambda)$ of P3HT:PC₆₁BM solar cells for different absorber thicknesses d_{PAL} . For ITO-free devices type A in **a** and ITO-based devices type C in **b**, the calculated radiant out-coupling factors $\bar{y}_{OC}(\lambda)$ for the corresponding layer stacks and the spectra corrected for the optical out-coupling $I_{EL,corr} = I_{EL} \cdot \bar{y}_{OC}^{-1}$

out that the PAL and contact materials were not changed and morphologic changes are not expected by mere deposition of such a thin Ag layer on top of the PEDOT:PSS layer and thus the recombination properties are also not expected to change. Accordingly, the different spectrum has to originate from the change of the optical properties of the system. In addition, the calculated radiant out-coupling factor \bar{y}_{OC} for the corresponding layer stacks are plotted in Fig. 2 (dashed lines). In case of device type B, the out-coupling factor \bar{y}_{OC} resembles the spectral shape quite accurately proving further that the spectrum is very strongly governed by out-coupling effects. Therefore, the data hardly allows for any statement about the spectral fingerprint of the emitting photoactive material, i.e., the energetics of the CT complex.

For type A, the wavelength dependence of \bar{y}_{OC} originates from the interference of the direct emission and the reflected portion at the back mirror, i.e., wide-angle interference. In this case, the path difference between direct and reflected light depends on the distance of the photon emission from the reflective electrode, thus, the averaging over all emission positions within the PAL leads to a quite homogeneous out-coupling. Still, aspects of the spectral shape like the asymmetry of the spectrum can be partially explained by optical out-coupling. In contrast, the additional highly reflective surface in the type B stack forms an optical microcavity leading to a pronounced peak in \bar{y}_{OC} (multibeam interference). While the amplitude of the out-coupled radiance $I_{PAL}(x)$ and thus $\bar{y}_{OC}(x)$ still depend on the position of emission within the cavity the peak position is determined by the optical distance of the mirrors.

The previously discussed case has a more exemplary character, since a reflective surface at the illumination side reduces photon absorption and hence is avoided in the design of an OSC. The following part therefore focuses on regularly used architectures of OSC. The EL spectra of P3HT:PC₆₁BM devices (type A) for varying PAL thicknesses d_{PAL} are shown in Fig. 3a along with the corresponding $\bar{y}_{OC}(\lambda)$ and the corrected spectra $I_{EL,corr} = I_{EL} \cdot \bar{y}_{OC}^{-1}$. The corrected intensity $I_{EL,corr}$ represents the expected free space radiance without the optical environment, i.e., the emission in a homogeneous medium. In Fig. 3b, the same quantities are plotted for ITO-based devices (type C in Fig. 1c).

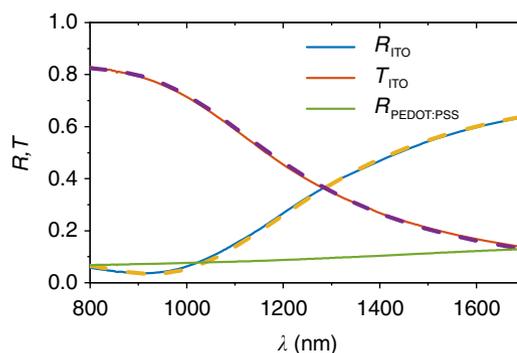


Fig. 4 Reflectance (R) and transmittance (T) of ITO and the reflectance of PEDOT:PSS. Materials were measured on glass (solid lines) and the fits (dashed lines) to the ITO data for the determination of n and k values. ITO shows a strong increase in reflectance for wavelengths longer than 1000 nm

The measured spectra depend sensitively on d_{PAL} and show peak positions ranging from 1240 to 1370 nm for device type A and from 1050 to 1380 nm for type C devices. In addition, peak amplitudes range from 3.4 to 15.3 for type A and from 1.5 to 23.8 for type C.

In great contrast, all-corrected intensities $I_{EL,corr}$ are quite similar regarding the shape and amplitude for the whole range of different values of d_{PAL} and for the two different device architectures as could be expected for the same photoactive material used in these devices. This is a clear indication that the measured spectra are strongly influenced by the optical properties of the layer stacks. The measured spectra using PEDOT:PSS as transparent contact (type A) are wider compared to the spectra of the ITO-based devices (type C), which show more narrow peaks. For both architectures, the overall intensities vary as a function of d_{PAL} but more severe for the ITO-based ones. While the out-coupling \bar{y}_{OC} in the PEDOT:PSS device is quite homogeneous as discussed previously, the out-coupling of the ITO-based devices feature clear peaks. These peaks originate from the high reflectance of ITO at the relevant wavelengths as shown in Fig. 4

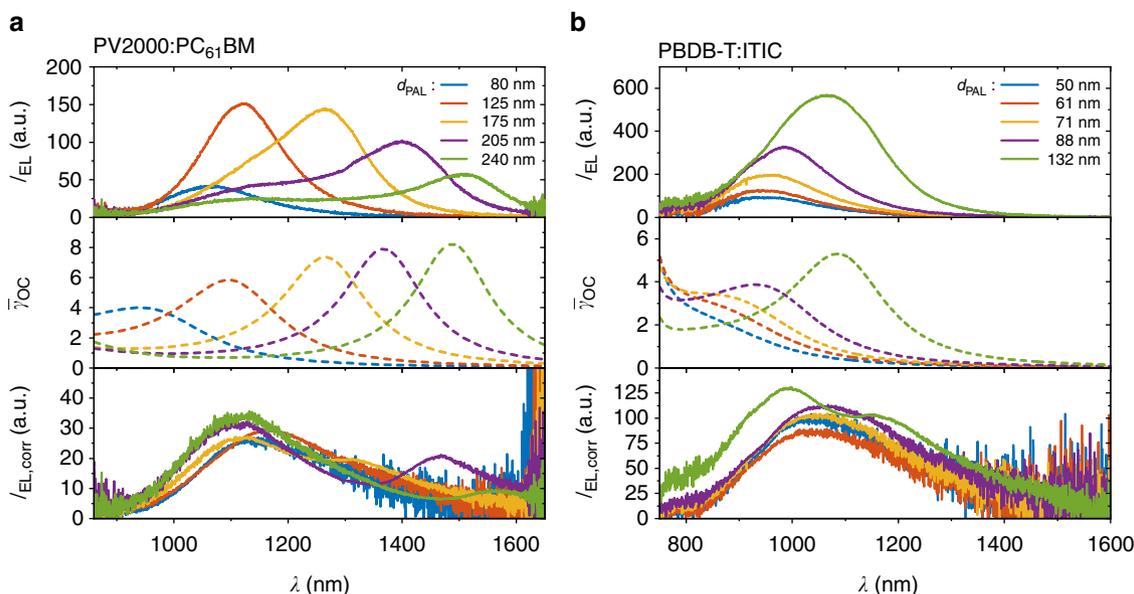


Fig. 5 Electroluminescence spectra for PV2000:PC₆₁BM and PBDB-T:ITIC solar cells. EL spectra $I_{EL}(\lambda)$ of PV2000:PC₆₁BM solar cells (**a**) and PBDB-T:ITIC solar cells (**b**) for different thicknesses d_{PAL} of the photoactive layer in an ITO-based design (type D in Fig. 1), the calculated radiant out-coupling factors $\bar{\gamma}_{OC}$ for the corresponding layer stacks and the spectra corrected for out-coupling

(measured on glass), resulting in the formation of an optical microcavity. In contrast, PEDOT:PSS (measured on glass) does not show a comparable increase of its reflectance in the infrared part of the spectrum (green curve in Fig. 4).

It should be noted that varying d_{PAL} can result in morphological and also energetic changes which may influence I_{EL} and thereby $I_{EL,corr}$ ³⁸. The deviations present in the corrected spectra are due to slight offsets in peak position and peak width between measurement and simulation. The results shown in Fig. 3 clearly demonstrate the strong impact of interference effects on the observed phenomena.

Additionally, the polymer PV2000 as an example for a high-performing polymer in combination with PC₆₁BM and the state of the art non-fullerene DA system PBDB-T:ITIC are investigated in ITO-based solar cells³⁹ (full names listed under cell preparation further below). This is shown in Fig. 1d (type D) yielding PCE between 4.9% ($d_{PAL} = 80$ nm) and 7.9% ($d_{PAL} = 240$ nm) in the case of PV2000:PC₆₁BM and 7.2% ($d_{PAL} = 50$ nm), 8.2% ($d_{PAL} = 132$ nm), and a maximum of 9.3% ($d_{PAL} = 88$ nm) for PBDB-T:ITIC. Electroluminescence spectra for the PV2000:PC₆₁BM devices are shown in Fig. 5a for layer thicknesses in the range of $80 \text{ nm} \leq d_{PAL} \leq 240$ nm alongside with $\bar{\gamma}_{OC}$ and $I_{EL,corr}$. While I_{EL} peaks between 1050 and 1500 nm with maximum peak intensities in the range of 40–150 (a.u.) all-corrected spectra feature a peak between 1120 and 1170 nm with maximal intensities between 26 and 34 (a.u.). In the uncorrected spectra, these devices show a shift in wavelength of the peak of almost 500 nm and a significant change in overall emission for different thicknesses of the absorber layer. This can clearly be ascribed as to originate from changes in the optical out-coupling properties upon variation of absorber thickness. The spectra I_{EL} , $I_{EL,corr}$ and $\bar{\gamma}_{OC}$ for the PBDB-T:ITIC devices for absorber thicknesses $50 \text{ nm} \leq d_{PAL} \leq 132$ nm are shown in Fig. 5b. Thicker photoactive layers could not be realized due to a poor resulting film quality. I_{EL} peaks between 940 and 1070 nm with maximum peak intensities in the range of 100–600 (a.u.), whereas all-corrected spectra $I_{EL} \cdot \bar{\gamma}_{OC}^{-1}$ feature a peak at ~ 1070 nm with maximal intensities between 85 and 12 (a.u.). Note that the strong s-shape in the corrected spectrum of the device with an average absorber thickness of 132 nm originates from strong fluctuations

of d_{PAL} of about 30 nm (as measured by profilometry), resulting in an inaccurate correction when only the average thickness is taken into account.

OSCs usually feature a reflective back contact. In case of no interference in the device, i.e., incoherent emission of light, γ_{OC} equals 2 since the radiation is reflected at the backside mirror and emitted only into one hemisphere. Hence, the total emitted power which is proportional to the square of the electric field of the wave $|E|^2$ remains unchanged. In contrast, for coherent light, $|E|$ is doubled when the criterion for constructive interference is fulfilled. For this reason, the radiance is enhanced by a factor of 4 compared to the free space emission thus resulting in a factor of 2 in total emitted power. As could be shown experimentally by Drexhage, this doubling of the total emitted power is accompanied by a decreased radiative lifetime for the emission⁴⁰. This decreased radiative lifetime for emission does also apply for the devices investigated in this study. However, the total recombination and thus the lifetime of electrons and holes in typical organic solar cells are strongly dominated by non-radiative recombination, which can clearly be seen from the previously mentioned low-quantum yields for radiative CT recombination in the range of only 10^{-9} – 10^{-6} in typical D–A systems. For this reason, interference effects do not have any measurable impact on the overall charge carrier lifetimes in these materials.

For the determination of the CT state energy, the previously shown spectra $I_{EL}(\lambda)$ (in $\text{W}/\text{nm}/\text{m}^2$) were converted to $I_t(E)$ given in photons per unit spectral energy and per area by dividing by the photon energy E ³⁴. The normalized reduced spectra I_t/E are plotted in Fig. 6 alongside with the corrected spectra $I_{t,corr}/E$ for all systems. The corrected spectra of PBDB-T:ITIC are smoothed by a floating average over 20 data points to reduce noise and to ensure proper normalization. Gaussian functions are fitted by means of Eq. (1) to all spectra and added to Fig. 6. The resulting CT state energy E_{CT} , reorganization energy λ_{RO} and the energy of maximum CT emission $E_{CT}^{em} = \max(I_t/E)$ from the fits for all systems and the corresponding photoactive layer thicknesses are listed in Table 1.

E_{CT}^{em} is strongly influenced by interference as previously seen from the wavelength-dependent spectra in Figs. 3 and 5. From Fig. 6 it gets obvious that the high energetic tails, i.e., the apparent

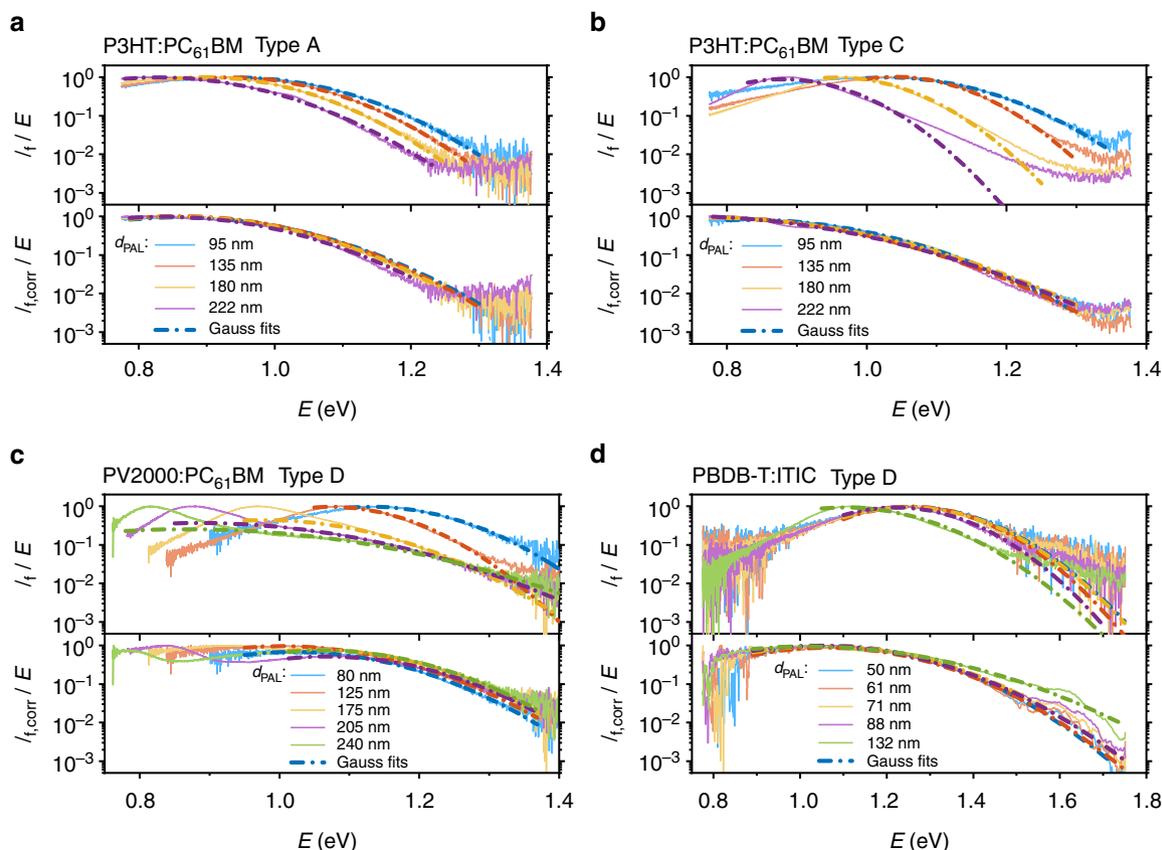


Fig. 6 Reduced emission spectra I_t/E and corrected spectra $I_{t,corr}/E$ data (solid lines). Gaussian fits (dashed lines) given by Eq. (1) for P3HT:PC₆₁BM devices of type A in **a** and P3HT:PC₆₁BM devices of type C in **b** as well as type D devices made from PV2000:PC₆₁BM (**c**) and PBDB-T:ITIC (**d**)

Table 1 Results of Gaussian fits to the experimental and corrected reduced EL spectra for different photoactive materials and varying absorber thickness d_{PAL} plotted in Fig. 6

d_{PAL} (nm)	E_{CT}^{em} (eV)	λ_{RO} (eV)	E_{CT} (eV)	$E_{CT,corr}^{em}$ (eV)	$\lambda_{RO,corr}$ (eV)	$E_{CT,corr}$ (eV)
P3HT:PC ₆₁ BM (type A)						
95	0.947	0.262	1.209	0.860	0.364	1.224
135	0.934	0.233	1.167	0.858	0.350	1.208
180	0.896	0.236	1.133	0.856	0.345	1.201
220	0.828	0.300	1.128	0.837	0.353	1.190
P3HT:PC ₆₁ BM (type C)						
95	1.028	0.227	1.256	0.830	0.388	1.218
135	1.040	0.131	1.171	0.764	0.493	1.256
180	0.957	0.131	1.089	0.762	0.540	1.302
220	0.880	0.127	1.007	0.727	0.594	1.321
PV2000:PC ₆₁ BM (type D)						
80	1.130	0.190	1.320	1.021	0.270	1.292
125	1.072	0.152	1.224	1.001	0.307	1.308
175	0.977	0.301	1.278	1.051	0.262	1.313
205	0.881	0.565	1.446	1.067	0.263	1.330
240	0.868	0.718	1.586	1.060	0.259	1.319
PBDB-T:ITIC (type D)						
50	1.251	0.353	1.604	1.067	0.586	1.653
61	1.254	0.308	1.562	1.062	0.620	1.682
71	1.249	0.351	1.600	1.056	0.666	1.723
88	1.222	0.316	1.538	1.039	0.696	1.736
132	1.101	0.455	1.556	0.997	1.168	2.166 ^a

^aValue not reliable due to layer thickness fluctuation (see text and Fig. 5b)

reorganization energy λ_{RO} , and thus the estimated CT energy $E_{CT} = E_{CT}^{em} + \lambda_{RO}$ are significantly affected by interference effects as well. The spectra of the ITO-based P3HT:PC₆₁BM and PBDB-T:ITIC devices (Fig. 6b–d) show that the uncorrected data sets differ from the corrected spectra for all absorber thicknesses and in case of P3HT:PC₆₁BM for $d_{PAL} > 135$ nm microcavity effects can result in observed spectra significantly deviant from the expected Gaussian shape. This results in a strong dependence of the fit results on the chosen data range making a fit of a Gaussian function questionable in general. In case of PV2000:PC₆₁BM (Fig. 6c), the uncorrected spectra look like a superposition of two peaks for $d_{PAL} \geq 175$ nm due to the resonant peak of the microcavity. These strongly altered spectral shapes make a meaningful analysis of the uncorrected data difficult and prone to misinterpretation. Interestingly, there are cases where two effects compensate each other when the CT state energy $E_{CT} = E_{CT}^{em} + \lambda_{RO}$ is determined by the applied fitting procedure. As an example, the uncorrected PBDB-T:ITIC spectra feature higher E_{CT}^{em} values with lower λ_{RO} (peak width) compared to the corrected spectra. This results in comparable CT energies although the values for E_{CT}^{em} and λ_{RO} differ significantly. Hence, even in this case the correction procedure is indispensable to achieve a deeper understanding of the underlying processes and material properties.

It can be seen in Table 1 that the corrected spectra resemble each other quite accurately for all absorber thicknesses and the fitted $E_{CT,corr}$ coincide within <0.1 eV. In contrast, the raw E_{CT} energies deviate up to more than 0.35 eV for the investigated systems (note that it could be even higher for other systems). This deviation should by no means be misinterpreted as an upper limit for the error of the CT state energy as quantities derived from

uncorrected spectra are not meaningful if interference effects are dominant. In general, cell architectures featuring ITO as transparent electrode show more pronounced microcavity effects due to the high reflectivity of ITO for wavelengths longer than 1000 nm as explained in detail before. The described effects are expected to be negligible if devices which do not differ significantly in terms of their optical properties are compared among each other as could be the case when different interlayers with comparable optical constants are compared or when mere morphological changes are induced by additives or annealing procedures. Nevertheless, consistent layer thicknesses should be ensured in such cases. In contrast, correction for optical effects is crucial if the actual spectral shape of the luminescence spectrum is of interest in organic thin-film devices.

It should be stated here, since out-coupling of light is ultimately the reverse process of light absorption, the results will be affected in a comparable manner if low energetic CT state absorption tails are analyzed.

Discussion

CT states are crucial for the thorough analysis of organic solar cells as their energy and occupation determine the open-circuit voltage (V_{OC}) and thus have a large impact on the achievable power conversion efficiency. For this reason, an exact determination of the CT state energy is of utmost importance. CT state energies are experimentally determined by different methods which however are always based—directly or indirectly—on absorption and emission of light. That the CT state emission can be largely influenced by the optical properties of the device in the corresponding wavelength region is shown in this work for a variety of different bulk heterojunction organic solar cells based on four different device architectures, three different photoactive materials, and various thicknesses of the absorber layer. For each device, an electroluminescence spectrum was recorded and it was found that the spectra for a given photoactive material differ strongly for different device architectures and different thicknesses of the absorber layer. In harsh contrast, the CT state energy as a fundamental property of the donor/acceptor blend is expected to be rather independent of the exact device configuration and the thickness of the absorber layer. Therefore, the optical out-coupling properties of the different devices were calculated by means of a scattering matrix formalism based on experimentally determined optical diffraction and extinction coefficients. After correction of the luminescence spectra for the specific out-coupling properties, all spectra for each photoactive material were very similar. These findings demonstrate clearly that the measured spectra are strongly influenced by microcavity effects and that the described correction for the optical out-coupling is indispensable in order to achieve reliable information about the nature of the emissive states within the photoactive layer. It is important to note that the described correction procedure is required for the correct analysis of CT state absorption spectra.

Methods

Optical constants. Reflection and transmission (RT) spectra were measured for the utilized materials using a PerkinElmer LAMBDA 950 UV/Vis Spectrophotometer equipped with an integrating sphere. The corresponding layer thicknesses were measured with a Veeco Dektak 150 surface profilometer. The refractive indices (n) and extinction coefficients (k) were determined by fitting the measured RT spectra using the software SCOUT from W. Theiss Hard- and Software. The values for Ag^{42} , MoO_3^{43} , and ZnO^{44} were obtained from literature. All n and k values used within this study are plotted in the Supplementary Figures 1, 2.

Cell preparation. Three different blend solutions for the photoactive layer were used. (i) Poly(3-hexylthiophen-2,5-diyl):[6,6]-phenyl-C61-butyric acid methyl ester (P3HT:PC₆₁BM) (25:17.5 mg/ml in *o*-dichlorobenzene), (ii) PV2000:PC₆₁BM

(12:18 mg/ml in *o*-xylene), and (iii) Poly[(2,6-(4,8-bis(5-(2-ethylhexyl)thiophen-2-yl)-benzo[1,2-b:4,5-b']dithiophene)-alt-(5,5-(1',3'-di-2-thienyl-5',7'-bis(2-ethylhexyl)benzo[1',2'-c:4',5'-c']dithiophene-4,8-dione)):3,9-bis(2-methylene-(3-(1,1-dicyanomethylene)-indanone))-5,5,11,11-tetrakis(4-hexylphenyl)-dithieno[2,3-d:2',3'-d']-s-indaceno[1,2-b:5,6-b']dithiophene)] (PBDB-T:ITIC) (10:10 mg/ml in 97% vol. chlorobenzene and 3% vol. 1-chloronaphthalene). The polymeric donor materials were purchased from (i) Riecke; (ii) Raynergy Tek; and (iii) Solarmer Energy. The fullerene derivatives were purchased from Solenne and the non-fullerene acceptor ITIC was purchased from Solarmer Energy. The materials used for the charge selective interlayers were 2.5% wt. zinc oxide (ZnO) nanoparticle solution (Avantama N-11) purchased from Avantama AG, diluted to the final concentration of 1% wt. in 2-propanol (99.5 anhydrous), and the poly(3,4-ethylenedioxythiophene) polystyrene sulfonate (PEDOT:PSS) formulations Clevis™ AI 4083 and Clevis™ FHC Solar purchased from Heraeus. The layer stack and the thicknesses of the different layers for the ITO-based P3HT:PC₆₁BM cells were glass/ITO (230 nm)/PEDOT:PSS (Clevis™ AI 4083, 30 nm)/P3HT:PC₆₁BM/LiF (0.3 nm)/Al (100 nm). In case of PV2000, the used architecture is glass/ITO (180 nm)/ZnO (25 nm)/PV2000:PC₆₁BM/MoO₃ (10 nm)/Ag (100 nm). As for PBDB-T, the used architecture is glass/ITO/ZnO (20 nm)/PBDB-T:ITIC/MoO₃ (2 nm)/Ag (100 nm). In these architectures, PEDOT:PSS solutions were spun cast at 3000 rpm and annealed at 130 °C for 15 min under nitrogen, ZnO diluted solutions were spun cast at 4000 rpm and annealed at 120 °C for 10 min under nitrogen. Different layer thicknesses were achieved by a variation in spin-coating speed. The photoactive layers were annealed under nitrogen at 130 °C for 15 min (P3HT:PC₆₁BM), at 80 °C for 2 min, 130 °C for 5 min, and 110 °C for 2 min (PV2000:PC₆₁BM), and 140 °C for 15 min (PBDB-T:ITIC). The ITO-free layer stack for P3HT:PC₆₁BM was glass/Cr (5 nm)/Al (100 nm)/Cr (5 nm)/P3HT:PC₆₁BM/PEDOT:PSS (Clevis™ FHC Solar, 70 nm)/Au (grid, 100 nm). For these samples, PEDOT:PSS solution was spun cast at 3000 rpm and annealed together with the photoactive layer at 150 °C for 10 min under nitrogen. LiF, MoO₃, Ag, Au, and Al were all thermally evaporated at a pressure below 5×10^{-6} mbar.

Luminescence spectroscopy. The setup consists of a Czerny-Turner spectrograph (Andor Shamrock 193i) with an InGaAs photo diode array (Andor iDus DU490A), which was cooled to -70 °C. Using a collimating system consisting of two lenses with an angle of acceptance of $\sim 10^\circ$ (full angle), the samples were imaged to the plane of the entrance slit of the spectrograph. A 850-nm-long pass color glass filter was used to exclude the presence of peaks originating from the second order of diffraction from the grating. All measured spectra were corrected for the spectral sensitivity of the setup using a tungsten halogen lamp as a reference standard (reference spectrum was measured by Fraunhofer ISE CalLab). Although the absolute intensities are unknown all measured spectra $I_{EL}(\lambda)$ have the same relative unit proportional to $W/m^2/nm$ and can be compared among each other. The working point of the OSC was controlled by a Keithley 2400 sourcemeter. EL spectra were measured at an applied forward current density of 75 mA/cm² (P3HT:PC₆₁BM), 108.1 mA/cm² (PV2000:PC₆₁BM), and 100 mA/cm² (PBDB-T:ITIC) while the samples were produced, stored, and measured in an inert nitrogen atmosphere. The width of the entrance slit of the spectrograph was 2 mm for P3HT:PC₆₁BM and 0.5 mm for PV2000:PC₆₁BM and PBDB-T:ITIC and the height of the detector array was 0.2 mm.

Code availability. The computer code used during the current study is not publicly available.

Data availability

The data that support the findings of this study are available from the corresponding author upon request.

Received: 23 March 2018 Accepted: 7 August 2018

Published online: 07 September 2018

References

1. Vandewal, K., Tvingstedt, K., Gadisa, A., Inganäs, O. & Manca, J. V. Relating the open-circuit voltage to interface molecular properties of donor:acceptor bulk heterojunction solar cells. *Phys. Rev. B* **81**, 125204 (2010).
2. Vandewal, K. Interfacial charge transfer states in condensed phase systems. *Annu. Rev. Phys. Chem.* **67**, 113–133 (2016).
3. Bredas, J.-L., Norton, J. E., Cornil, J. & Coropceanu, V. Molecular understanding of organic solar cells: the challenges. *Acc. Chem. Res.* **42**, 1691–1699 (2009).
4. Deibel, C., Strobel, T. & Dyakonov, V. Role of the charge transfer state in organic donor-acceptor solar cells. *Adv. Mater.* **22**, 4097–4111 (2010).
5. Würfel, U., Thorwart, M. & Weber, E.R. (eds) *Quantum Efficiency in Complex Systems, Part II—From Molecular Aggregates to Organic Solar Cells* (Elsevier, New York, 2011).

6. Tvingstedt, K. et al. Electroluminescence from charge transfer states in polymer solar cells. *J. Am. Chem. Soc.* **131**, 11819–11824 (2009).
7. Burke, T. M., Sweetnam, S., Vandewal, K. & McGehee, M. D. Beyond Langevin recombination. How equilibrium between free carriers and charge transfer states determines the open-circuit voltage of organic solar cells. *Adv. Energy Mater.* **5**, 1500123 (2015).
8. Synooka, O. et al. Influence of thermal annealing on PCDTBT:PCBM composition profiles. *Adv. Energy Mater.* **4**, 1300981 (2014).
9. Tvingstedt, K., Vandewal, K., Zhang, F. & Inganäs, O. On the dissociation efficiency of charge transfer excitons and frenkel excitons in organic solar cells: a luminescence quenching study. *J. Phys. Chem. C* **114**, 21824–21832 (2010).
10. Spies, A., List, M., Sarkar, T. & Würfel, U. On the impact of contact selectivity and charge transport on the open-circuit voltage of organic solar cells. *Adv. Energy Mater.* **7**, 1700855 (2017).
11. Loi, M. A. et al. Charge transfer excitons in bulk heterojunctions of a polyfluorene copolymer and a fullerene derivative. *Adv. Funct. Mater.* **17**, 2111–2116 (2007).
12. Hallermann, M., Haneder, S. & Da Como, E. Charge-transfer states in conjugated polymer/fullerene blends. Below-gap weakly bound excitons for polymer photovoltaics. *Appl. Phys. Lett.* **93**, 53307 (2008).
13. Wetzelaer, G.-J. A. H., Kuik, M. & Blom, P. W. M. Identifying the nature of charge recombination in organic solar cells from charge-transfer state electroluminescence. *Adv. Energy Mater.* **2**, 1232–1237 (2012).
14. Vandewal, K., Tvingstedt, K., Gadisa, A., Inganäs, O. & Manca, J. V. On the origin of the open-circuit voltage of polymer-fullerene solar cells. *Nat. Mater.* **8**, 904–909 (2009).
15. Kim, H. et al. Electroluminescence in polymer-fullerene photovoltaic cells. *Appl. Phys. Lett.* **86**, 183502 (2005).
16. Wittmann, H. F. et al. Microcavity effect in a single-layer polymer light-emitting diode. *Adv. Mater.* **7**, 541–544 (1995).
17. Puzzo, D. P. et al. Organic light-emitting diode microcavities from transparent conducting metal oxide photonic crystals. *Nano Lett.* **11**, 1457–1462 (2011).
18. Xiang, L., Fuxiang, W. & Hui, L. Spectrum study of top-emitting organic light-emitting devices with micro-cavity structure. *J. Semicond.* **30**, 44007 (2009).
19. Meerheim, R., Furno, M., Hofmann, S., Lüssem, B. & Leo, K. Quantification of energy loss mechanisms in organic light-emitting diodes. *Appl. Phys. Lett.* **97**, 253305 (2010).
20. Granlund, T., Pettersson, L. A. A., Anderson, M. R. & Inganäs, O. Interference phenomenon determines the color in an organic light emitting diode. *J. Appl. Phys.* **81**, 8097 (1997).
21. Salinas, J.-F. et al. Optical design of transparent thin metal electrodes to enhance in-coupling and trapping of light in flexible polymer solar cells. *Adv. Mater.* **24**, 6362–6367 (2012).
22. Inganäs, O., Tang, Z., Bergqvist, J. & Tvingstedt, K. in *Organic Solar Cells* (eds Rand, B. & Richter, H.) 367–427 (Pan Stanford Publishing, Singapore, 2014).
23. Sessler, M. *Selektive Fensterschichten in organischen Tandemsolarzellen*. PhD thesis, Albert-Ludwigs-Universität Freiburg im Breisgau (2015) <https://doi.org/10.6094/UNIFR/10626>.
24. Sergeant, N. P. et al. Design of transparent anodes for resonant cavity enhanced light harvesting in organic solar cells. *Adv. Mater.* **24**, 728–732 (2012).
25. Lin, H.-W. et al. Device engineering for highly efficient top-illuminated organic solar cells with microcavity structures. *Adv. Mater.* **24**, 2269–2272 (2012).
26. Long, Y. Improving optical performance of inverted organic solar cells by microcavity effect. *Appl. Phys. Lett.* **95**, 193301 (2009).
27. Yu, W. et al. Effects of the optical microcavity on the performance of ITO-free polymer solar cells with WO₃/Ag/WO₃ transparent electrode. *Sol. Energy Mater. Sol. Cells* **100**, 226–230 (2012).
28. Huang, Z.-Y. et al. Spontaneous formation of light-trapping nano-structures for top-illumination organic solar cells. *Nanoscale* **6**, 2316–2320 (2014).
29. Zuo, L. et al. Microcavity-enhanced light-trapping for highly efficient organic parallel tandem solar cells. *Adv. Mater.* **26**, 6778–6784 (2014).
30. Chen, Y.-H. et al. Microcavity-embedded, colour-tunable, transparent organic solar cells. *Adv. Mater.* **26**, 1129–1134 (2014).
31. Vandewal, K., Tvingstedt, K. & Inganäs, O. in *2011—Quantum Efficiency in Complex Systems* (eds Würfel, U et al.) 261–295 (Elsevier, New York, 2011).
32. Marcus, R. A. Relation between charge transfer absorption and fluorescence spectra and the inverted region. *J. Phys. Chem.* **93**, 3078–3086 (1989).
33. Bödicker, F.M.J. Bachelor thesis. Technische Universität Darmstadt (2012).
34. Li, L. Formulation and comparison of two recursive matrix algorithms for modeling layered diffraction gratings. *J. Opt. Soc. Am. A* **13**, 1024–1035 (1996).
35. Whittaker, D. M. & Culshaw, I. S. Scattering-matrix treatment of patterned multilayer photonic structures. *Phys. Rev. B* **60**, 2610–2618 (1999).
36. Jackson, J.D. *Classical Electrodynamics* 3rd edn (Wiley, Hoboken, NY, 2009–2013).
37. Hofmann, C. L. M., Herter, B., Fischer, S., Gutmann, J. & Goldschmidt, J. C. Upconversion in a Bragg structure. Photonic effects of a modified local density of states and irradiance on luminescence and upconversion quantum yield. *Opt. Express* **24**, 14895–14914 (2016).
38. van Bavel, S., Sourty, E., With, G., de, Frolic, K. & Loos, J. Relation between photoactive layer thickness, 3D morphology, and device performance in P3HT/PCBM bulk-heterojunction solar cells. *Macromolecules* **42**, 7396–7403 (2009).
39. Zhao, W. et al. Fullerene-free polymer solar cells with over 11% efficiency and excellent thermal stability. *Adv. Mater.* **28**, 4734–4739 (2016).
40. Drexhage, K. H. Influence of a dielectric interface on fluorescence decay time. *J. Lumin.* **1–2**, 693–701 (1970).
41. Mooney, J. & Kambhampati, P. Get the basics right. Jacobian conversion of wavelength and energy scales for quantitative analysis of emission spectra. *J. Phys. Chem. Lett.* **4**, 3316–3318 (2013).
42. McPeak, K. M. et al. Plasmonic films can easily be better. Rules and recipes. *ACS Photonics* **2**, 326–333 (2015).
43. Lajaunie, L., Boucher, F., Dessapt, R. & Moreau, P. Strong anisotropic influence of local-field effects on the dielectric response of α -MoO₃. *Phys. Rev. B* **88**, 115141 (2013).
44. Query, M.R. *Optical Constants*. Report No. D-A158-623 (University of Missouri, 1985).

Acknowledgements

M.L. acknowledges the support from the Dr. Ruth Heerdt-Stiftung. The authors thank Florina Bödicker for the implementation of the S-matrix code and Oliver Höhn for guidance during the optical simulations. P.P. and J.A. received funding from the European Union's Horizon 2020 research and innovation program under the Marie Skłodowska-Curie Grant Agreement No. 713750. They further acknowledge support of the Regional Council of Provence-Alpes-Côte d'Azur, A*MIDEX (No. ANR-11-IDEX-0001-02), and the Investissements d'Avenir project funded by the French Government, managed by the French National Research Agency (ANR). U.W. acknowledges financial support by the German Federal Ministry for Economic Affairs and Energy under contract number 0324214G.

Author contributions

The research idea and design of experiments for its confirmation were made by M.L. and U.W. The optimization of the PBDB-T:ITIC device configuration was carried out by P.P. under the supervision of J.A. Most solar cells used for this study were built by C.L. Electroluminescence experiments were performed by M.L. and T.S., the optical simulations were carried out by M.L. The data were analyzed by M.L. and the manuscript was written by M.L. and U.W.

Additional information

Supplementary Information accompanies this paper at <https://doi.org/10.1038/s41467-018-05987-8>.

Competing interests: The authors declare no competing interests.

Reprints and permission information is available online at <http://npg.nature.com/reprintsandpermissions/>

Publisher's note: Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.



Open Access This article is licensed under a Creative Commons Attribution 4.0 International License, which permits use, sharing, adaptation, distribution and reproduction in any medium or format, as long as you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons license, and indicate if changes were made. The images or other third party material in this article are included in the article's Creative Commons license, unless indicated otherwise in a credit line to the material. If material is not included in the article's Creative Commons license and your intended use is not permitted by statutory regulation or exceeds the permitted use, you will need to obtain permission directly from the copyright holder. To view a copy of this license, visit <http://creativecommons.org/licenses/by/4.0/>.

© The Author(s) 2018