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# Thickness-Dependent Optical Nonlinearities of Nanometer-Thick $\text{Sb}_2\text{Te}_3$ Thin Films: Implications for Mode-locking and Super-Resolved Direct Laser Writing

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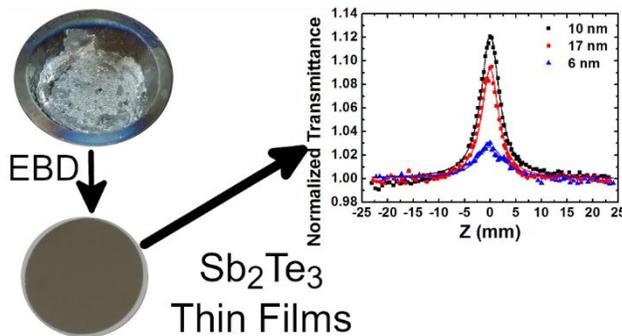
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**ABSTRACT:** 2D materials are currently very promising candidates for various photonic applications. Optimizing their optical nonlinearities requires a thorough adjustment of several properties including the film thickness. In this work thin  $\text{Sb}_2\text{Te}_3$  layers with different thicknesses (ranging from 2.5 nm to 50 nm) are prepared by the electron beam deposition technique, then they are properly annealed in order to achieve significant third order nonlinearities. The film structure and morphology are extensively studied by means of X-Ray Diffraction, Scanning Electron Microscopy, Transmission Electron Microscopy and Energy-dispersive X-Ray spectroscopy. The presence of nanocrystals with sizes highly depending on the film thickness has been observed. Optical studies are carried out by Vis-NIR spectrophotometric studies. Finally, a thickness dependent study of the ultrafast third order nonlinear optical properties of the  $\text{Sb}_2\text{Te}_3$  thin films is performed. This study is carried out by means of the Z-scan technique, employing 400 fs laser pulses at 1030 nm. The observed optical nonlinearities are very high, compared with those of state-of-the-art nonlinear optical materials. Moreover they are highly dependent on the thickness of the layers. The findings demonstrate the importance of a fine adjustment of the  $\text{Sb}_2\text{Te}_3$  thickness in order to enhance its nonlinear optical efficiency. They are expected to be of significant importance for mode-locking of laser systems and super-resolved direct laser writing.

Scheme:



## INTRODUCTION

After the successful isolation of graphene layers and the intense research carried out concerning their nonlinear optical properties, 2D materials are very widely investigated in order to be used for photonic applications. This is currently a field of very intense scientific research and many publications are already existing.<sup>1-13</sup> Several 2D materials exhibit unique optical and nonlinear optical (NLO) properties which are mainly due to their topological insulating characters.<sup>4,14</sup> More specifically, these materials have an insulating character in the bulk, while conducting states appear at the surface. This results to a band structure which is similar to that of graphene, showing a Dirac-like linear band dispersion.<sup>15</sup> Such materials are currently the most promising candidates for applications requiring significant NLO responses and in several cases it has been shown that they can have superior performances compared to those of graphene.<sup>5,16</sup> Moreover, the broadband character of their optical and nonlinear optical responses enhances their importance for photonic applications.<sup>3,17</sup>

The relation between the thickness of 2D materials and their nonlinearities is still unclear, despite their importance for many different fields of photonics and the intense research currently carried out in this scientific domain. This is a significant scientific problem nowadays which requires more and thorough studies. **Shedding light on** this aspect will provide strong enhancements to the efficiency of 2D materials in terms of applications. Research in this field has been significantly triggered by the rise of graphene, for this reason it is the mostly investigated material in this domain.

Among 2D materials, the  $\text{Sb}_2\text{Te}_3$  is well known for its very high nonlinear optical responses. More specifically, it has been shown in the past, that it exhibits a very high saturable absorption

at the ns regime.<sup>18-20</sup> This has been employed in order to surpass the diffraction limit and increase the information density of optical data storage media.<sup>18</sup> Due to their topological insulator character,  $\text{Sb}_2\text{Te}_3$  thin films can have broadband optical nonlinearities. Indeed our group has demonstrated that such films can exhibit significant NLO responses at the visible and the IR parts of the spectrum (532 nm and 1064 nm respectively) using nanosecond laser pulses.<sup>17</sup> Very recently we have demonstrated the same broadband character at the ultrafast regime employing 600 fs duration laser pulses at both 515nm and 1030 nm.<sup>16</sup> More specifically, a very high saturable absorption has been obtained, which has been compared with current state-of-the-art materials.<sup>16</sup>

Due to its high saturable absorption, the  $\text{Sb}_2\text{Te}_3$  material is very widely used nowadays for mode-locking of ultrafast laser systems.<sup>21-24</sup> Despite of the importance of this material for photonic applications, its nonlinear optical parameters are not sufficiently investigated.

An enhanced NLO response will improve the efficiency of currently used mode-locking systems and will provide novel ways to obtain super-resolved nanostructuring at the ultrafast regime. Indeed, the fabrication of efficient mode-locking materials requires a significantly high saturable absorption. The same NLO mechanism can result to the decrease of the size of a focused Gaussian beam. This can offer sub-wavelength resolutions during nanostructuring of a material with femtosecond pulses. In order to enhance the NLO efficiency of  $\text{Sb}_2\text{Te}_3$  thin films thorough studies of the optical nonlinearities are required. These studies are highly complex because they require a deep understanding of the laser-matter interaction of this material under femtosecond pulse irradiation. The difficulty is further increased, due to the fact that enhancing the NLO response requires a well-defined crystalline structure. This has been previously achieved by our group by means of a precise annealing of the  $\text{Sb}_2\text{Te}_3$  thin layers.<sup>16</sup> The enhancement of the

optical nonlinearities is also expected to depend on the nanocrystal size. Moreover, taking into account that the significant nonlinearities of  $\text{Sb}_2\text{Te}_3$  thin films are attributed to their topological insulator character, a high dependence of the NLO response on the thin film thickness can be expected. Such studies have been carried out in the case of other materials, such as graphene and resulted to an enhancement of their NLO performances.<sup>25</sup> It has to be noted that there is currently no evidence concerning the relation of the nonlinear absorption and nonlinear refraction parameters of the  $\text{Sb}_2\text{Te}_3$  material with the film thickness and the nanocrystal size at the ultrafast regime.

The target of this work is to shed light upon this matter and trigger further research in this field. The article is structured as follows. Firstly the thin film preparation is presented. Then the results obtained by X-Ray Diffraction (XRD), Scanning Electron Microscopy (SEM), Transmission Electron Microscopy (TEM), Energy-Dispersive X-Ray Spectroscopy (EDS) and Vis-IR spectroscopy studies are shown. All the aforementioned studies are of significant importance because they allow a precise and complete identification of the structure, morphology and optical properties of the layers. This is necessary in order to relate the NLO properties with the thin film structure and shed light upon the complex laser-matter interactions in the case of the  $\text{Sb}_2\text{Te}_3$  material. Finally, the NLO properties are investigated by means of the Z-scan technique employing 400 fs laser pulses at 1030 nm. The results are compared with those previously obtained in the literature for other 2D photonic materials.

## RESULTS AND DISCUSSION

**Thin films preparation.**  $\text{Sb}_2\text{Te}_3$  thin films have been deposited on 2 mm thick B270 substrate using Electron Beam Deposition (EBD) by a Bühler SYRUSpro 710 machine, at room

temperature with a  $0.5 \text{ nmsec}^{-1}$  deposition rate. A previously calibrated quartz microbalance has been employed during the thin film deposition in order to measure the thickness of the  $\text{Sb}_2\text{Te}_3$  layers with a precision of  $\pm 0.2 \text{ nm}$ . The  $\text{Sb}_2\text{Te}_3$  thin films thicknesses are ranging from 2.5 nm to 50 nm. The nanometer-thick  $\text{Sb}_2\text{Te}_3$  thin films have been then protected by a 2 nm thick  $\text{SiO}_2$  layer in order to avoid their oxidation after the deposition.

A proper annealing of the thin films is necessary in order to obtain a high nonlinearity, as the latter is highly dependent on the crystal structure.<sup>16</sup> The crystallization has been obtained by performing a 24 hour annealing at  $300^\circ\text{C}$  as previously demonstrated by our group.<sup>16,17</sup>

**Morphological studies and elemental analysis.** The annealed layers have been initially visualized by Scanning Electron Microscopy using a backscatter electron detector that delivers a signal response from deeper subsurface layers of the sample and also shows crystal orientation contrast.<sup>26</sup> SEM images were acquired at 10 kV acceleration voltage, which was the optimal one to observe the  $\text{Sb}_2\text{Te}_3$  surface between the silica protection layer and the substrate. Figures 1a-c, acquired at the same magnification for comparison, show nanocrystals with different sizes and contrasts, revealing their polycrystalline nature. Additionally, the observation of the total area shows that the crystallization occurs all over the films. Figure 1a corresponds to the 10 nm thick film, exhibiting nanocrystal length size ranging from 15 nm to 127 nm, with an average length of 46 nm. In Figures 1b and 1c SEM images of the 30 nm and the 50 nm thick layers are shown, respectively. Figure 1b shows nanocrystal sizes ranging from 30 to 310 nm with an average length of 104 nm, whereas in Figure 1c the nanocrystal size varies from 46 to 447 nm, with an average length of 164 nm.

To further investigate the film growth and perform elemental analysis, a Scanning Transmission Electron Microscopy (STEM) study coupled to Energy-dispersive X-ray spectroscopy analyses

have been performed. For this study, the samples were first carbon coated to prevent any charging effects, and ultrathin electron transparent cross sections were prepared by Focused Ion Beam (FIB) milling. In Figures 2a,b cross-sectional bright field STEM images of the 10 nm and 30 nm thick  $\text{Sb}_2\text{Te}_3$  films are presented, respectively. The annealed surfaces have been found to be slightly rough and the film thicknesses were in very good agreement with the nominal thicknesses. EDS spectra were acquired to check the chemical composition of the layers. (see scanned zones in Figures 2a and 2b). The focused electron beam was in the range of a few nanometers, allowing to probe only the layer and prevent any contribution from either the glass substrate or the protective silicon oxide layer on top. The obtained spectra are presented in Figures 2c and 2d for the two different sample thicknesses. The concentration deduced from the Sb (K) and Te (K) ionization peaks is respectively equal to 37 and 63 atomic percents in very good agreement with the nominal concentrations of the  $\text{Sb}_2\text{Te}_3$  thin films. For this measurement, K lines were preferred to L lines, as they are less absorbed by the sample itself.

**X-Ray diffraction studies.** X-Ray diffraction studies have been performed for all annealed films studied in this work. The studies have been carried out with a theta-theta configuration and  $\text{CuK}\alpha$  radiation ( $\lambda=0.15418$  nm) using an X'Pert MPD diffractometer, equipped with a rapid detector. The representative diagrams shown in Figures 1d-f have been obtained for the same thin films used for the SEM studies (10 nm, 30 nm, 50 nm samples, Figures 1a-c) to facilitate the comparison of the results. The XRD peaks obtained belong to the rhombohedral crystalline structure  $\overline{R3}m$  with 00l preferred orientation of the  $\text{Sb}_2\text{Te}_3$  material, in accordance with the results previously obtained by our group and other scientific groups.<sup>17,16,27-31</sup> In the case of the 7 nm thick layers and for lower film thicknesses no peaks appeared during the XRD studies. These

films are though well crystallized (see Figure S1 in Supporting information for the 7 nm thick layer). We attribute this to a very low size of the crystallized grains which are under the detection level of the XRD technique. It has been also found that the XRD peaks were becoming narrower by increasing the film thickness. This indicates that the nanocrystals formed during the annealing become larger as the film thickness increases which is in very good agreement with the SEM studies. It is important to note that with the theta-theta configuration used, the families of probed diffracted planes are parallel to the substrate. Thus, the information deduced from the diffracted peak is only linked to these planes. It is the case for the crystal sizes that have been determined by means of the Scherrer equation<sup>32</sup> for each film and are shown in Table 1. It can be seen that the determined crystal sizes are similar to the thickness of the films, which demonstrates that the crystallization took place everywhere between the substrate and the silica protection layer, on the totality of the thickness.

**Optical studies.** Vis-NIR spectrophotometric studies have been performed by means of a Perkin-Elmer Lambda 1050 with a specific module that allows performing the reflectance and transmittance measurement at 8° angle of incidence on the same point of a sample. The measurements have been performed from 400 nm to 1200 nm. Representative transmittance and reflectance curves are presented in Figures 3a, 3b for 6 different film thicknesses. The obtained curves indicate the broadband optical characteristics of the Sb<sub>2</sub>Te<sub>3</sub>, which are due to its low bandgap (between 0.14 eV and 0.20 eV).<sup>33,34</sup> In Figure 3c the reflectance, the transmittance and the absorption at the laser excitation wavelength used for the NLO studies (i.e. 1030 nm) are shown as a function of the film thickness for all the thin films investigated. Films with thicknesses up to 5 nm exhibit transmittances higher than 60%. The transmittance gradually decreases with increasing the film thickness. This is mainly due to a significant increase of the

reflectance of the thin film layers as the thickness increases. Indeed a very high reflectance, of about 70% has been obtained in the case of the 30 nm thick  $\text{Sb}_2\text{Te}_3$  film. The absorption of the thin films has been found to increase with increasing the thin film thicknesses. An absorption maximum of about 35% has been obtained for the 8 nm thick  $\text{Sb}_2\text{Te}_3$  layer. A slight, gradual decrease of the absorption has been obtained for films thicker than 11 nm. This can be attributed to the fact that the crystal sizes and morphology highly depend on the film thickness, as shown in the previous paragraphs, which can have a high impact on the optical properties of the investigated layers.

The refractive index of all annealed  $\text{Sb}_2\text{Te}_3$  layers was determined in the range [400-1200 nm]. The technique followed, based on spectrophotometric analysis, is the curve fitting method, which requires an optical dispersion model. This model is capable of integrating several spectral information, such as transmittance and reflectance, allowing the index determination process. In addition, it is also capable of handling complicated cases, such as strongly absorbing layers.

In this case, the complex refractive index of annealed  $\text{Sb}_2\text{Te}_3$  films of different thicknesses is estimated from the reflectance and transmittance values. The theoretical reflectance and transmittance were calculated using the matrix method, taking into account backside reflectance. They depend on the measurement wavelength, the substrate refractive index  $n_s$ , the complex refractive index  $(n, k)$  and the layer thickness  $(d)$ . There are several optical dispersion models describing the laws of the complex refractive index of the layer under consideration. In this study, we focused on a Forouhi-Bloomer model combined with a Drude model.<sup>35</sup>

Several considerations justify this choice. The Forouhi-Bloomer model is one of the optical dispersion models derived from Kramers-Kronig integration. All the parameters describing the real and imaginary parts of the refractive index have a physical meaning. Finally, this model

shows excellent performance with some crystalline and amorphous semiconductors, but also with dielectric thin films over a wide energy range.

The determination of the index consists of minimizing an error function, denoting the difference between the calculated and experimental reflectance or transmittance values in the whole spectral domain, and as a function of the 8 Forouhi Bloomer and Drude parameters injected on the  $n$  and  $k$  model.

Of course, with a relatively large number (8) of variables needed to calculate the refractive index  $n$  and  $k$ , and thus the error function, a simple local non-linear least squares optimization technique is inefficient, and a global optimization procedure is needed. As indicated in our previous study,<sup>35</sup> a global clustering optimization (GCO) algorithm is very efficient for the determination of a multiparametric index. GCO methods can be considered as an advanced form of a standard Multistart procedure, in which a local search is performed from several starting points distributed over the entire search initial domain.

The real and imaginary parts of the refractive index, corresponding to those leading to the overall minimization of the error function, are shown in Figures 3d,e. The validity and accuracy of index determination increases with the thickness. Indeed, we consider a homogeneous layer, which generally cannot be guarantee for very thin layers, in the 0-5nm range. Moreover, spectral signatures are not specific in the case of very thin layers, and several  $n$  and  $k$  dispersion curves may give a good agreement. Regarding our results, index determination seems reliable for samples with thicknesses higher than 10 nm, but should be considered carefully for thinner layers.

However, the values determined in this work are in good agreement with the values previously reported in the literature.<sup>36</sup> In Figure 3f, the same parameters are presented at 1030 nm as a function of the layer thickness.

**Nonlinear optical studies.** For the needs of the nonlinear optical studies the Z-scan technique has been employed (see also experimental section). The studies have been performed using 400 fs duration pulses, at 1030 nm with a 100 Hz repetition rate. Many different thin films (annealed and amorphous) exhibiting thicknesses from 2.5 nm to 30 nm have been studied during this work. Higher thicknesses have not been studied in order to avoid low laser beam transmittances. It has to be noted that in all cases the amorphous layers have been found to exhibit negligible nonlinearities, compared with the annealed ones (see for example Figure S2 in the supporting information), showing the importance of the annealing to obtain high NLO responses. The results shown in the next paragraphs concern the annealed Sb<sub>2</sub>Te<sub>3</sub> layers. The substrates used for the thin film deposition have been separately studied, under identical experimental conditions, revealing that they do not contribute to the nonlinear refraction and the nonlinear absorption of the investigated structures. Many Z-scan studies have been performed for every thin film and in many different areas of the surface of the samples. Moreover measurements have been done for many different laser intensities in order to find the most suitable range of intensities to be employed for this work and verify that no modification of the thin film layers was occurring during the nonlinear optical investigations. The range of intensities used for the Z-scan studies was 0.4 GW/cm<sup>2</sup> to 10 GW/cm<sup>2</sup>, which is significantly lower than the damage threshold, which is about 20 GW/cm<sup>2</sup> under the same experimental conditions.

In Figure 4a representative normalized “open aperture” Z-scan curves obtained with a  $6 \text{ GW/cm}^2$  incident laser intensity are presented. For film thicknesses lower than 5 nm the obtained signals were found to be very low corresponding to negligible optical nonlinearities. For thicknesses higher than 5 nm a transmission peak has been observed (see for example the obtained experimental curve for the 6 nm thin film), indicating the saturable absorption character of the  $\text{Sb}_2\text{Te}_3$  layers. The obtained peaks were gradually increasing with the film thickness and a maximum peak has been obtained for the 9 nm - 11 nm thick  $\text{Sb}_2\text{Te}_3$  layers. In Figure 4a a representative “Open aperture” Z-scan obtained for the 10 nm layer is presented. Then a decrease of the nonlinearity has been observed for thicker films. In Figures S3 and S4 (Supporting information) additional “Open aperture” curves obtained for thin films of different thicknesses are presented. All the curves have been recorded using the same incident laser intensity ( $6 \text{ GW/cm}^2$ ). Moreover in Figure S5 (Supporting information) three curves obtained at three different zones of the 10 nm thin film layer are presented, revealing the excellent repeatability of the measurements.

The nonlinear absorption coefficient ( $\beta$ ), as well as the imaginary part of the third order nonlinear susceptibility ( $\text{Im}\chi^{(3)}$ ) have been determined by many experimental curves, following the procedure described in the experimental section. Table 1 presents the nonlinear optical parameters for all thin films investigated during this work. The  $\text{Im}\chi^{(3)}$  is also presented in Figure 5a as a function of the thin film thickness. In Figure 5b the figure of merit, defined as the  $\text{Im}\chi^{(3)}$  divided by the linear absorption coefficient of the thin film layers at 1030 nm, is also presented. It is obvious that the imaginary part of the third order nonlinear susceptibility and the figure of merit obtain their maximum values for a film thickness of 9-11 nm. The nonlinear absorption gradually decreases for thicker and thinner films. This finding is in accordance with the

experimental curves shown in Figure 4a. It has to be noted that the choice of the thin film thicknesses employed for this investigation have been gradually done during the studies in order to allow a high precision determination of the highest nonlinear absorption that can be obtained by the  $\text{Sb}_2\text{Te}_3$  material.

In order to study the nonlinear refraction of the investigated materials “closed aperture” and “divided” Z-scans have been recorded. The determination of the nonlinear refraction of the  $\text{Sb}_2\text{Te}_3$  layers is a very challenging procedure for the following reasons. For thin films thicker than 6 nm the nonlinear absorption exhibits giant values and consequently it is dominating the optical nonlinearities. In such cases the  $\text{Re}\chi^{(3)}$  becomes much lower than  $\text{Im}\chi^{(3)}$  and it cannot be detected by the Z-scan technique.<sup>37</sup> Decreasing the thickness can aid to surpass this difficulty, however very low thicknesses result to negligible nonlinearities (both nonlinear refraction and nonlinear absorption). For these reasons the nonlinear refractive parameters of the  $\text{Sb}_2\text{Te}_3$  material at the ultrafast regime are unknown. These measurements can be accurately done for 5-6 nm thick layers. This is due to the fact that these layers exhibit low nonlinear absorption, allowing a precise determination of the nonlinear refraction. A representative “divided” Z-scan obtained for the 5 nm thin film layer with an incident laser intensity of  $6 \text{ GW/cm}^2$  is presented in Figure 4b. The “divided” Z-scan curves obtained during at this work show a peak-valley configuration, which indicates a self-defocusing character and consequently negative  $\text{Re}\chi^{(3)}$  values. The nonlinear refractive parameters have been determined in the case of the 5 and 6 nm layers following the procedure shown in the experimental section and are shown in Table 1.

Similar thickness dependent studies of the  $\text{Sb}_2\text{Te}_3$  material do not currently exist, despite the importance of this material for applications. Our work demonstrates the significant dependence of both nonlinear refraction and absorption on the thickness of the 2D layers. By a careful

examination of Table 1 one can identify five different regimes, which result to different NLO behaviours. The first regime corresponds to very thin films (i.e. 2.5 nm) which exhibit negligible nonlinear refraction and nonlinear absorption. The lack of high optical nonlinearities can be attributed to the fact that in this case there is not sufficient matter to provide a uniform crystallization of the layers. A second, different behaviour has been observed in the case of the 5 nm thick films. In that case there is a significant nonlinear refraction, while the thin films exhibit a negligible nonlinear absorption. The nonlinear refractive parameter ( $\gamma'$ ) and  $\text{Re}\chi^{(3)}$  obtained in this work are of the order of  $10^{-9} \text{ cm}^2/\text{W}$  and  $10^{-7} \text{ esu}$  respectively. These values are significantly high, showing that the  $\text{Sb}_2\text{Te}_3$  material can exhibit a strong nonlinear refractive character if its thickness is properly adjusted. It is interesting to make a comparison with the nonlinear refraction values previously obtained for graphene layers, as they are currently a reference 2D material for nonlinear optics. The published nonlinear optical properties of graphene vary a lot due to an apparent strong dependence on the pulse duration, the excitation wavelength, the repetition rate and the graphene preparation conditions. Chen *et al.*<sup>38</sup> reported a nonlinear refractive index of the order of  $1.4 \times 10^{-9} \text{ cm}^2/\text{W}$ , which is slightly higher than the values reported in this work. The repetition rate employed by Chen *et al.* has been high (i.e. 80 MHz). In another work similar values have been found ( $10^{-7} \sim 10^{-6} \text{ esu}$ ) by employing the same repetition rate.<sup>39</sup> Very recently Karampitsos *et al.*<sup>40</sup> have investigated graphene dispersions and have demonstrated that the repetition rate of the laser plays a very important role on the NLO response of graphene. Using a high repetition rate (i.e. 80 MHz) they have shown that thermal effects dominate the nonlinearities. To avoid this fact the repetition rate has been kept low (i.e. 100 Hz) during our work. Demetriou *et al.*<sup>41</sup> investigated the optical nonlinearities of few-layer graphene (5 to 7 layers) by wavelength-dependent measurements in the infrared. In order to assure that no

thermal effects arise they used a repetition rate of 1 kHz. The nonlinear refractive index obtained using an excitation wavelength at 1150 nm is of the same order of magnitude with our reported values for the  $\text{Sb}_2\text{Te}_3$  layers. It has to be noted that the negative nonlinear refraction obtained here (self-defocusing) is in accordance with several previous experimental and theoretical studies on graphene systems.<sup>40-42</sup> A negative nonlinear refraction has been also presented in Reference<sup>43</sup> in the case of another 2D material, black phosphorus, while the nonlinear refraction is 6-7 orders of magnitude lower than that reported in this work.

A third regime can be observed in Table 1 for the 6 nm layers. In this case there is a simultaneous existence of a nonlinear refraction and absorption. By further increasing the 2D layer thickness a transition to a fourth regime occurs, where the nonlinear absorption reaches record values, while gradually the nonlinear refraction disappears. This happens for thin films of 9-11 nm thickness. The nonlinear absorption coefficient has been found to be about an order of magnitude higher than that published by Liu *et al* for the same material.<sup>44</sup> The higher values obtained here can be attributed to an enhancement of the nonlinear absorption achieved through the thickness optimization that we present in this work and an annealing optimization that has been previously detailed by our group.<sup>16</sup> It has to be mentioned that the nonlinear absorption coefficients found in this work are slightly lower than those that we have previously reported, which is due to the shorter pulse duration of the laser system employed here (400 fs instead of 600 fs used in Reference<sup>16</sup>). The pulse duration is well-known to have a significant impact on the NLO response of 2D materials. For example Tang et al. have shown that a decrease of the pulse duration from 1 ps to 100 fs can result to a decrease of the nonlinear absorption coefficient of black phosphorus dispersions by about one order of magnitude.<sup>45</sup> The giant nonlinear absorption obtained for the annealed  $\text{Sb}_2\text{Te}_3$  thin films is among the highest ever reported. A

detailed comparison and discussion of these nonlinear optical absorption efficiencies with current state-of-the-art materials such as other chalcogenides,<sup>46,47</sup> perovskites,<sup>48–50</sup> graphene,<sup>51,52</sup> carbon nanotubes<sup>53</sup> and 2D  $\alpha$ -Mo<sub>2</sub>C<sup>54</sup> has been carried out by our group in Reference <sup>16</sup> (see Table 1 therein). The nonlinear optical mechanism which gives rise to these giant  $\beta$  values is the following. The Sb<sub>2</sub>Te<sub>3</sub> material exhibits a very low bandgap of about 0.14-0.20 eV, which offers to this material a broadband absorption character.<sup>16,33,34</sup> The excitation wavelength employed in this work (i.e. 1030 nm) excites the electrons from the valence band to the conduction band. Increasing the laser intensity results to an increase of the number of carriers in the conduction band. Since electrons are fermions, when all energy states are occupied, the Sb<sub>2</sub>Te<sub>3</sub> material is not able to absorb more light due to the Pauli blocking. This bleaching results to an increase of the transmission and leads to high NLO responses. The significant contribution of this mechanism to the NLO response of 2D materials has been previously demonstrated by our and other scientific groups.<sup>1,4,16</sup>

Finally a fifth regime can be defined for thicker films. An increase of the film thickness over 10 nm results to a gradual decrease of the nonlinear absorption coefficients. For thicknesses over 24 nm lower optical nonlinearities are observed, while the optical losses of the layers become significant.

The high dependence of the NLO response on the film thickness is attributed to the topological insulator behaviour of the thin film layers. Even slight modifications of the optimal thicknesses can result to a significant loss of the NLO performances. There is no evidence of this thickness dependence in the literature so a further comparison with the literature is not feasible. However thickness dependent studies have been performed for other 2D materials. Ahn *et al.* have recently investigated the optical nonlinearities of mono and multilayer graphene by means of Z-scan and

Optical Kerr gate experiments showing a high dependence on the number of graphene layers.<sup>39</sup> Chen *et al.* studied the nonlinear refraction and the nonlinear absorption from monolayers to multilayers in the case of graphene in the ultrafast regime.<sup>38</sup> They demonstrated that the nonlinear optical response is not proportional to the number of layers. This has been attributed to the interlayer coupling. In another work it has been demonstrated that monolayer graphene acts as a more effective saturable absorber than multilayer graphene.<sup>55</sup> More specifically the monolayer graphene has been found to provide a better pulse shaping ability, pulse stability and output energy. On the contrary, as we show in this work, in the case of the  $\text{Sb}_2\text{Te}_3$  material higher thicknesses are needed to obtain giant optical nonlinearities. This is related to the different nature of these two 2D materials. Graphene is obtained through the isolation of a single sheet of graphite and it already has the necessary structure to obtain the NLO response. In the case of the  $\text{Sb}_2\text{Te}_3$  material, formation of nanocrystals is necessary to achieve high nonlinear absorption or nonlinear refraction, which is obtained through a precise annealing of the deposited layers. Bao *et al.*<sup>25</sup> demonstrated that the saturation intensity of graphene systems can vary from 0.71 to 0.61 MW  $\text{cm}^{-2}$  by changing the number of layers from  $3 \pm 1$  to  $10 \pm 1$ . They also found that the modulation depth can be reduced from 66.5% to 6.2% by changing the number of graphene layers. Other 2D materials have been also previously investigated. Thickness-dependent studies have been also carried out in the case of  $\text{SnS}_2$  nanolayers in order to optimize their performance for fibre lasers.<sup>56</sup> In this case three different thicknesses have been studied and an increase of the modulation depth with increasing the film thickness has been found, while the best performance has been acquired by a 108.7 nm thick layer. In another work the nonlinear refraction and the nonlinear absorption has been studied as a function of the size of black phosphorus nanosheets.<sup>43</sup> Three different thicknesses have been studied showing an increase of both real and imaginary

part of the third order nonlinear optical susceptibilities by decreasing the nanosheets size. Tuo et al studied 2D  $\alpha$ -Mo<sub>2</sub>C crystals and find out that by changing the crystal thickness and the laser intensity a significant tunability of the NLO parameters can be achieved.<sup>54</sup>

The results reported in this work are of significant importance for the field of nonlinear optics of 2D materials. They show that a precise tuning of the film thickness is necessary in order to enhance the optical nonlinearities of the Sb<sub>2</sub>Te<sub>3</sub> layers. On the contrary, avoiding such adjustment can result to poor performances in terms of applications, as for example mode-locking of fibre lasers. Moreover the thickness adjustment can result to a switching between different NLO behaviours, so it can be employed in order to adapt the Sb<sub>2</sub>Te<sub>3</sub> performance for specific needs. More specifically, careful adjustment of the crystallized Sb<sub>2</sub>Te<sub>3</sub> thickness can provide a switching from a purely nonlinear refractive NLO regime to a regime where the nonlinear absorption dominates the nonlinearities. Another important point has to be highlighted. The nonlinear optical parameters found in this work demonstrate a non-monotonic dependence on the film thickness. Such behaviour can only be detected by thorough NLO studies of many different thin films. It is very possible that a similar behaviour will be unveiled for many other 2D materials by investigating a large number of samples, having appropriately adjusted thicknesses.

## CONCLUSIONS

The nonlinear optical properties of nanometer-thick 2D Sb<sub>2</sub>Te<sub>3</sub> thin films have been studied by the Z-scan technique employing 400 fs laser pulses at 1030 nm with a low repetition rate of 100 Hz in order to avoid the appearance of thermal effects. Significant NLO properties have been found, which are highly dependent on the film thickness. A nonlinear refractive parameter of the

order of  $10^{-9}$  cm<sup>2</sup>/W has been obtained for 5-6 nm thick layers. This refraction is very high, comparable with that of other state-of-the-art nonlinear optical materials. A giant nonlinear absorption coefficient of the order of  $10^{-6}$  m/W has been obtained for 9-11 nm thick films. The nonlinear absorption dominates in that case the optical nonlinearities. **These results show the importance of the precise adjustment of the film thickness for mode locking and super-resolved nanostructuring.** We believe that these findings will trigger further research in the field of the optical nonlinearities of 2D photonic materials.

## METHODS

**Sample Characterization.** SEM analysis was performed with a Zeiss GeminiSEM 500 ultra-high resolution Field Emission Scanning Electron Microscope operating at 10 kV. A backscatter electron detector was used for imaging in nanoscale. Grain sizes were measured using the Image J software.

For TEM analysis, the samples were observed with a FEI Tecnai G2 microscope operating at 200KV. This microscope is equipped with an X-Max Silicon Drift Detector (Oxford) which allows Energy-dispersive X-ray spectroscopy measurements. The Cliff-Lorimer method as implemented in the Aztec software package from Oxford is used for quantitative chemical analysis.

**Nonlinear Optical Studies.** The nonlinear optical measurements have been performed by means of the Z-scan technique. The Z-scan technique<sup>37</sup> employs two different experimental arms, the so called “closed aperture” and “open aperture” Z-scans allowing a determination of the nonlinear refraction and the nonlinear absorption of a material respectively. **For the Z-scan studies an Amplitude, Yuja, femtosecond laser system has been employed. This is a hybrid (crystal/fiber), passively mode-locked laser delivering 400 fs duration pulses at 1030 nm. The oscillator**

provides pulses at a 40 MHz repetition rate. By means of a pulse picker and an acousto-optic modulator, both integrated in the laser system, the repetition rate can be adjusted. For the Z-scan measurements a 100 Hz repetition rate has been employed in order to avoid thermal effect contribution to the measured nonlinear optical responses. The laser beam has been focused by a 10 cm focal length lens on the sample, which has been moved around the focal plane by means of a computer controlled translation stage. The beam waist at the focal plane has been measured to be 20  $\mu\text{m}$ .

The recorded “Closed aperture” and “Open aperture” Z-scans allow the determination of the real and imaginary parts of the third order nonlinear optical susceptibility ( $\text{Re}\chi^{(3)}$  and  $\text{Im}\chi^{(3)}$ ) respectively. The nonlinear absorption coefficient  $\beta$  has been determined by fitting the “open aperture” Z-scan curves by means of the following equation:

$$T = \frac{1}{\sqrt{\pi} \left( \frac{\beta I_0 L_{\text{eff}}}{1+z^2/z_0^2} \right)} \int_{-\infty}^{\infty} \ln \left[ 1 + \frac{\beta I_0 L_{\text{eff}}}{1+z^2/z_0^2} \exp(-t^2) \right] dt \quad (1)$$

where  $I_0$  is the on-axis irradiance at the focus,  $L_{\text{eff}}$  is defined as  $L_{\text{eff}}=(1-\exp(-\alpha_0 L))/\alpha_0$  and  $\alpha_0$  is the linear absorption coefficient of the sample (at 1030 nm and at 515 nm respectively). The  $\text{Im}\chi^{(3)}$  can be then calculated by means of the following equation:

$$\text{Im} \chi^{(3)}(\text{esu}) = \frac{10^{-7} c^2 n_0^2}{96 \pi^2 \omega} \beta(\text{cm W}^{-1}) \quad (2)$$

where  $c$  is the speed of light in  $\text{cm s}^{-1}$ ,  $n_0$  is the linear refractive index and  $\omega$  is the fundamental frequency in cycles per sec.

The nonlinear refraction parameters have been obtained by dividing the “closed aperture” Z-scans by the “open aperture” Z-scans, obtaining the so-called “divided” Z-scans. The “divided” Z-scans allowed the determination of the nonlinear refractive parameters following the

experimental analysis procedure previously detailed.<sup>57</sup> The  $\gamma'$  values have been determined as follows:

$$\gamma' = \frac{\sqrt{2}}{k I_0 L_{\text{eff}} 0.406(1-S)^{0.25}} \Delta T_{P-V} \quad (3)$$

where  $k$  is the wave number in vacuum,  $S$  is the linear transmittance of the aperture and  $\Delta T_{P-V}$  is the difference between the transmittance minima and maxima of the recorded experimental curves.

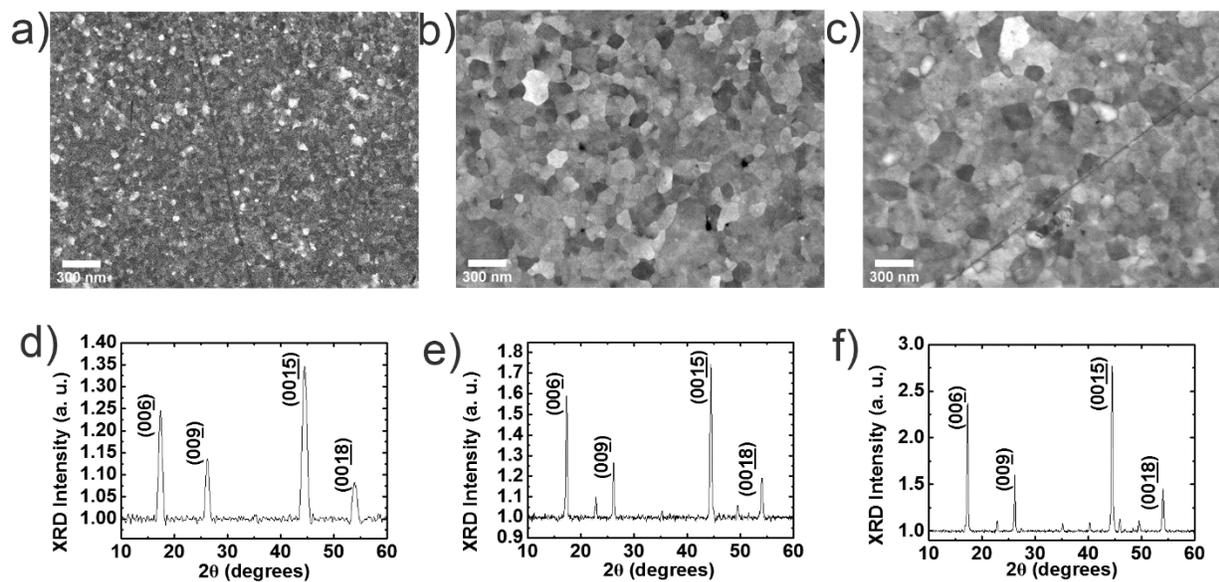
The real part of the third order nonlinear optical susceptibility has been then obtained by means of the following equation:

$$\text{Re } \chi^{(3)}(\text{esu}) = \frac{10^{-6} c \gamma'}{480 \pi^2} \quad (4)$$

where  $c$  is the speed of light in  $\text{cm s}^{-1}$  and  $\gamma'$  has to be given in  $\text{cm}^2 \text{W}^{-1}$ .

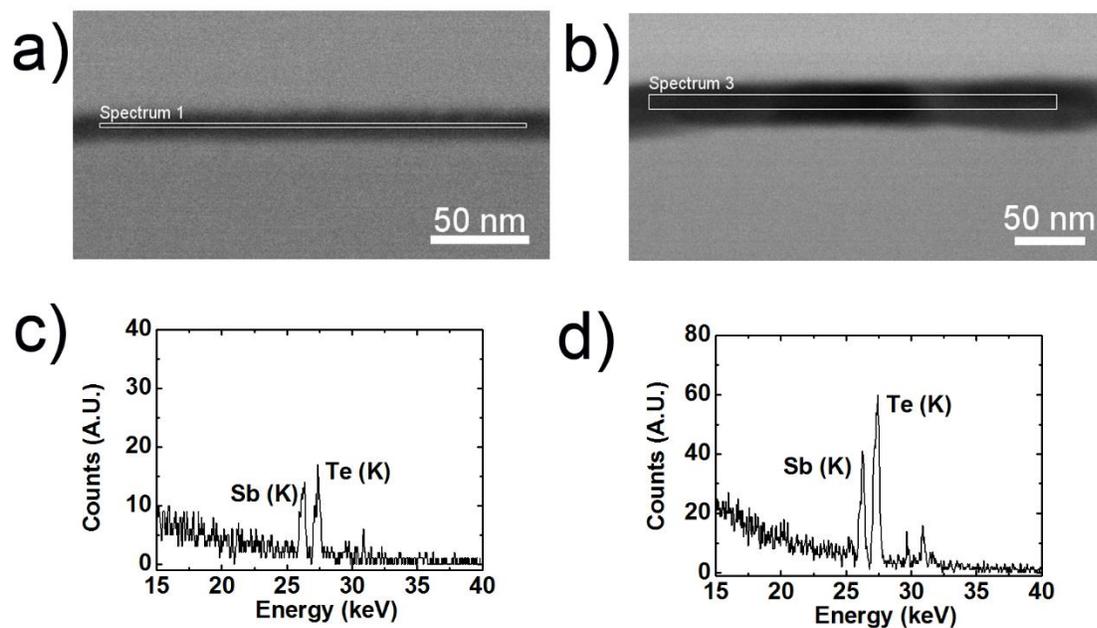
## FIGURES

### Figure 1



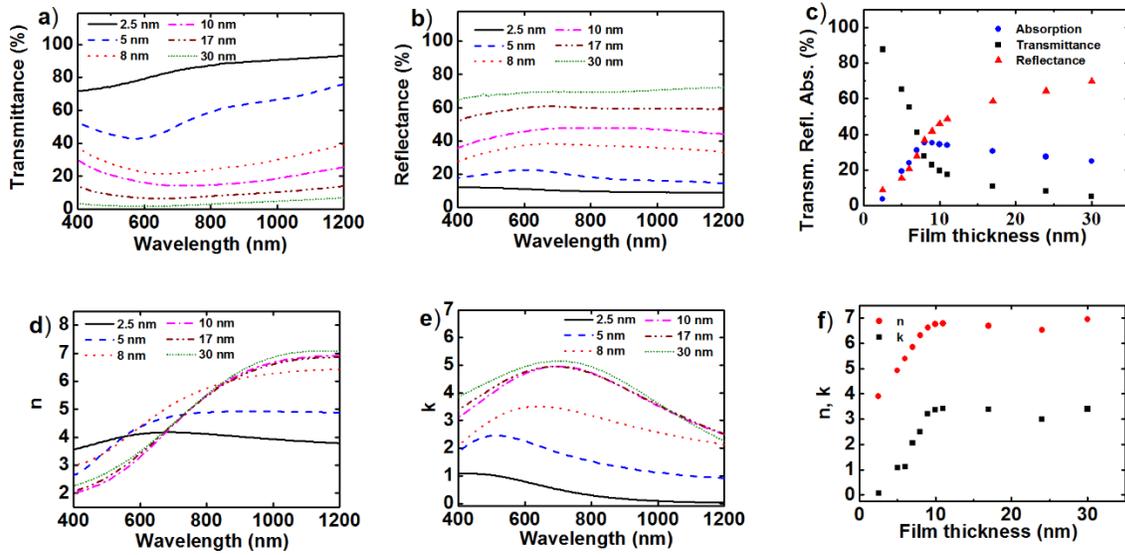
**Figure 1.** a-c) SEM images obtained in backscattered mode for the a) 10 nm b) 30 nm and c) 50 nm  $\text{Sb}_2\text{Te}_3$  layers. d-f) Corresponding XRD diagrams ( $\lambda = 0.15418$  nm) for the d) 10 nm e) 30 nm f) 50 nm samples.

**Figure 2**



**Figure 2.** Cross-sectional STEM bright field images for the a) 10 nm and b) 30 nm annealed  $\text{Sb}_2\text{Te}_3$  layers. The rectangle indicates the area scanned by the beam for the EDX analysis. EDX Spectra obtained for the c) 10 nm and d) 30 nm samples. The Sb(K) and Te(K) ionization peaks used for quantitative analysis are located at 26.4 keV and 27.4 keV.

**Figure 3**



**Figure 3.** a) Transmittance and b) Reflectance spectra for different film thicknesses. c) Transmittance, reflectance and absorption at 1030 nm as a function of the  $\text{Sb}_2\text{Te}_3$  thickness. Spectral dependence of the d) Real and e) Imaginary parts of the refractive index for different film thicknesses. f) Real and imaginary parts of the refractive index at 1030 nm as a function of the film thickness for all the investigated layers.

Figure 4

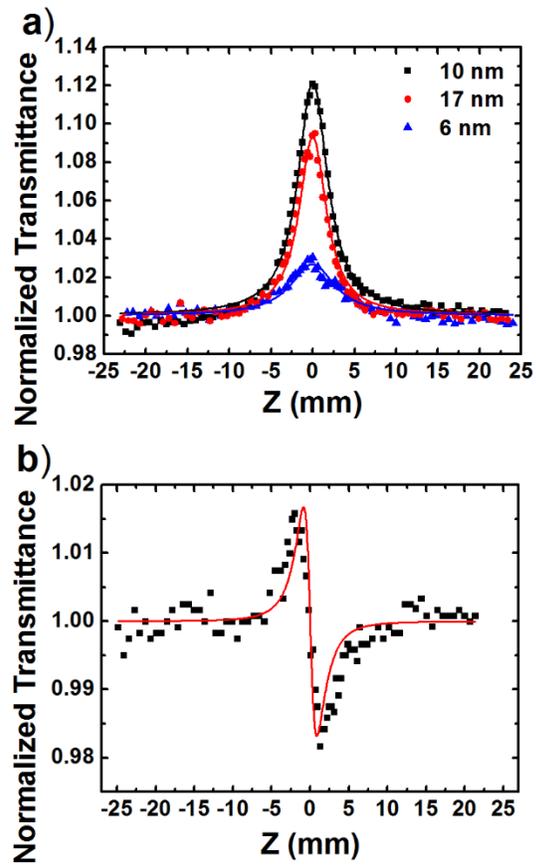
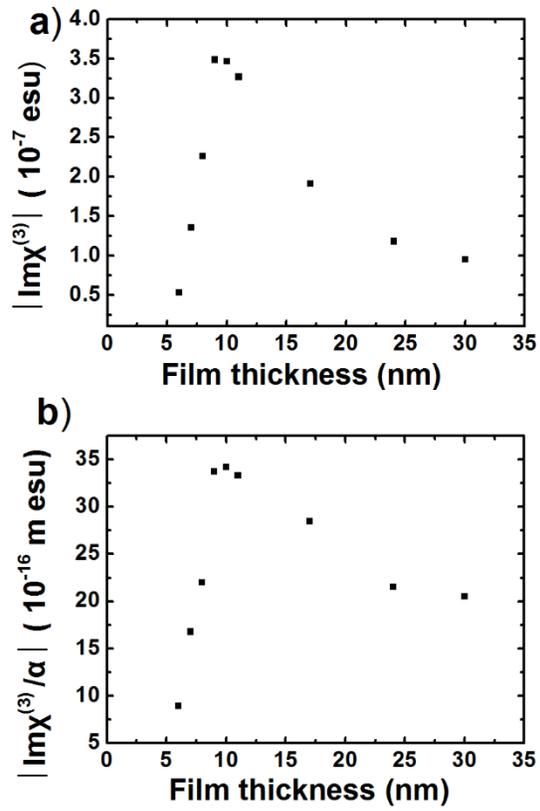


Figure 4. a) "Open aperture" Z-scan curves for three different  $\text{Sb}_2\text{Te}_3$  thicknesses. b) "Divided" Z-scan obtained for the 5 nm thick layer.

**Figure 5**



**Figure 5.** a)  $|\text{Im}\chi^{(3)}|$  and b) Figure of merit as a function of the thin film thickness.

## TABLES

**Table 1.** Determined nonlinear optical parameters (1030 nm, 400 fs)

| Film thickness (nm) | Crystal sizes (nm) | $\beta$ ( $10^{-7} \text{ mW}^{-1}$ ) | $\text{Im}\chi^{(3)}$ ( $10^{-7} \text{ esu}$ ) | $\text{Im}\chi^{(3)}/\alpha$ ( $10^{-16} \text{ m esu}$ ) | $\gamma'$ ( $10^{-10} \text{ cm}^2\text{W}^{-1}$ ) | $\text{Re}\chi^{(3)}$ ( $10^{-7} \text{ esu}$ ) |
|---------------------|--------------------|---------------------------------------|---|---|--|---|
| 2.5                 | N/A                | N/A                                   | N/A   | N/A   | N/A  | N/A   |
| 5                   | N/A                | N/A                                   | N/A   | N/A   | -8.70  | -1.33   |
| 6                   | N/A                | -3.54                                 | -0.53   | -8.90   | -11.24   | -2.05   |
| 7                   | N/A                | -7.62                                 | -1.35   | -16.78  | N/A  | N/A   |
| 8                   | N/A                | -10.90                                | -2.26   | -22.02  | N/A  | N/A   |
| 9                   | 10±2               | -15.30                                | -3.49   | -33.74  | N/A  | N/A   |
| 10                  | 11±2               | -14.59                                | -3.46   | -34.20  | N/A  | N/A   |
| 11                  | 12±2               | -13.65                                | -3.27   | -33.30  | N/A  | N/A   |
| 17                  | 18±2               | -9.62                                 | -2.24   | -28.47  | N/A  | N/A   |
| 24                  | 27±2               | -5.96                                 | -1.32   | -21.52  | N/A  | N/A   |
| 30                  | 31±2               | -4.79                                 | -1.21   | -20.52  | N/A  | N/A   |

## ASSOCIATED CONTENT

### Supporting Information.

SEM image of a 7 nm  $\text{Sb}_2\text{Te}_3$  layer. “Open aperture” Z-scan curves of an amorphous and annealed layer (same thickness). “Open aperture” Z-scans of different thickness  $\text{Sb}_2\text{Te}_3$  layers. Repeatability of the Z-scan measurements.

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### Author Contributions

R.N.V. and C.M performed the nonlinear optical studies. F.L. carried out the determination of the refractive index of the thin films. A.C. and M.C. performed the SEM, TEM and EDX characterizations and analysed the results. C.P.P performed the X-Ray Diffraction studies and analysed the experimental findings. J.L. deposited the thin film layers and performed the spectrophotometric studies. J.Y.N. and K.I. supervised the project and wrote the manuscript with input from all authors.

## NOTES

The authors declare no competing financial interest.

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TOC Image:

$$\beta \sim 10^{-7} \text{ mW}^{-1} \quad \gamma' \sim 10^{-9} \text{ cm}^2 \text{ W}^{-1}$$

